

杂原子掺杂闪蒸石墨烯 Heteroatom-Doped Flash Graphene

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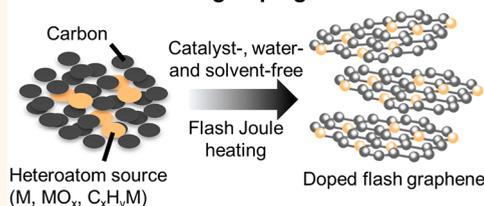
Supporting Information

摘要: 杂原子掺杂可以有效地调整本征二维材料的局部结构和电子态,并赋予其改进的光学、电学和机械性能。

ABSTRACT: Heteroatom doping can effectively tailor the local structures and electronic states of intrinsic two-dimensional materials, and endow them with modified optical, electrical, and mechanical properties. Recent studies have shown the feasibility of preparing doped graphene from graphene oxide and its derivatives via some post-treatments, including solid-state and solvothermal methods, but they require reactive and harsh reagents. However, direct synthesis of various heteroatom-doped graphene in larger quantities and high purity through bottom-up methods remains challenging. Here, we report catalyst-free and solvent-free direct synthesis of graphene doped with various heteroatoms in bulk via flash Joule heating (FJH). Seven types of heteroatom-doped flash graphene (FG) (boron, nitrogen, oxygen, phosphorus, sulfur), two-element-co-doped FG (boron and nitrogen), as well as three-element-co-doped FG (boron, nitrogen, and sulfur), are synthesized using various dopants, such as elements, oxides, and organic compounds are used. The graphene quality of heteroatom-doped FG is high, and similar to intrinsic FG, the material exhibits turbostraticity, increased interlayer spacing, and superior dispersibility. Electrochemical oxygen reduction reaction of different heteroatom-doped FG is tested, and sulfur-doped FG shows the best performance. Lithium metal battery tests demonstrate that nitrogen-doped FG exhibits a smaller nucleation overpotential compared to Cu or undoped FG. The electrical energy cost for the synthesis of heteroatom-doped FG synthesis is only 1.2 to 10.7 kJ/g, which could render the FJH method suitable for low-cost mass production of heteroatom-doped graphene.

KEYWORDS: flash Joule heating, direct synthesis, flash graphene, heteroatom-doping, catalyst-free

Flash Joule heating doping



INTRODUCTION

本征纳米材料的化学功能化已广泛用于调整局部结构和电子状态,从而极大地影响材料的光学、电学和机械性能。Chemical functionalization of intrinsic nanomaterials has been widely used to tailor the local structures and electronic states, thereby greatly affecting the materials' optical, electrical, and mechanical properties. Traditional methods to achieve chemical modification involve the introduction of various heteroatoms,^{3,5} specific functional groups,^{1,6} or even other molecules⁷ into the intrinsic lattice. Heteroatom doping, which consists of substituting the original components with other elements, is the most basic yet highly effective modification.^{5,8} For graphene, many different dopants can be introduced, including nitrogen, sulfur, oxygen, boron, phosphorus, and some metal atoms. Heteroatom-doped graphene has demonstrated electrocatalytic properties,⁹ superior performance in sensing applications,⁵ and stronger interactions with various polymers as well as single metal atoms.¹⁰ Previous research has focused on the chemical functionalization of preformed graphene lattice (post-treatment), including ball milling,⁵ thermal annealing of graphene oxide (GO),^{13,14} plasma treatment, and photochemical reactions.¹⁵ Ball milling and plasma methods are simple and scalable routes to synthesize heteroatom-doped graphene, but the doping is usually limited at the edges or other defects.^{5,15,16} Thermal

热退火方法不仅需要GO作为在剧烈氧化下合成的前驱体,还需要延长高温以部分恢复sp²碳网络。annealing method not only needs GO as the precursor, which is synthesized under harsh oxidation, but also requires extended high temperature to partially recover the sp² carbon network. These top-down insertions into graphene usually cause irreversible degradation of graphene quality, and these procedures can be problematic since they result in inevitable impurities from the excessive supply of reactive agents such as KMnO₄ and BCl₃.^{1,5} Most bottom-up direct synthesis of heteroatom-doped graphene can be achieved by chemical vapor deposition (CVD)¹⁷ or synthetic organic strategies,¹⁸ but the products are limited to ultrasmall amounts, from μg- to mg-scales, and they often suffer from catalytic metal impurities such as Cr, Ti, and Ni. Furthermore, the types of heteroatom-doped graphene compositions achievable are often restricted by the vapor pressure or solubility of the precursors.⁵

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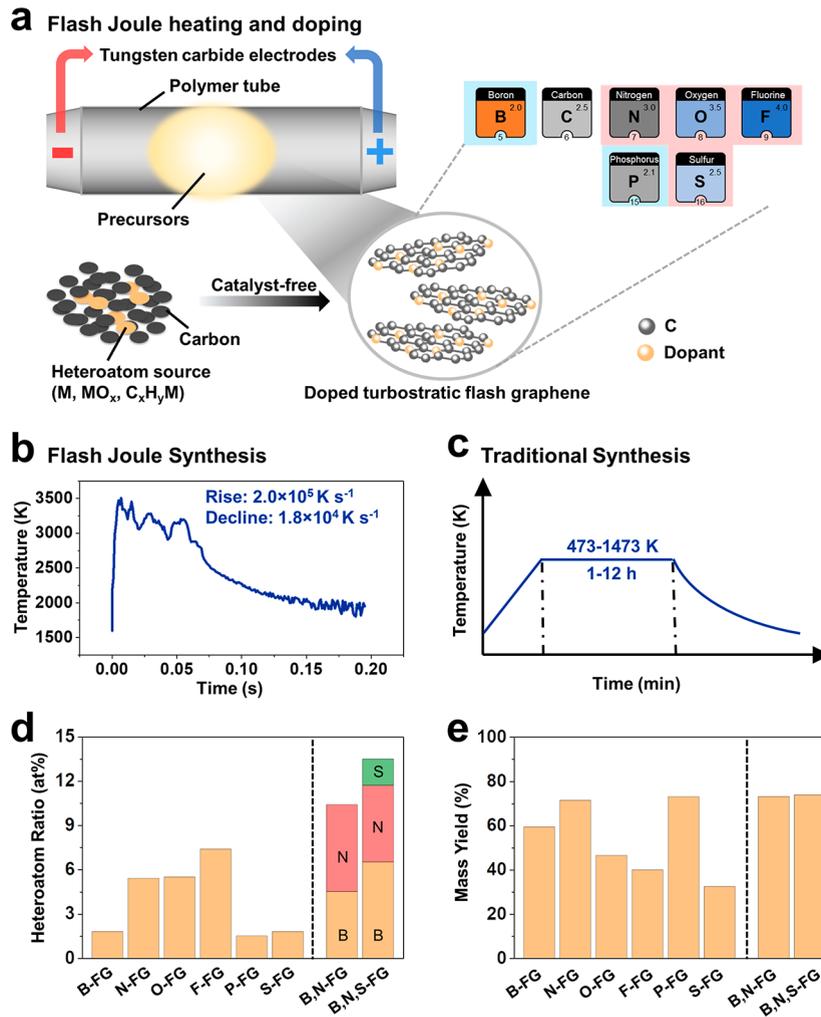


图1. 闪焦热直接合成掺杂原子的石墨烯。 (a) 通过闪焦热直接合成掺杂原子掺杂层石墨烯的示意图。 (b) 掺杂闪蒸石墨烯直接FJH合成过程中的实时温度测量。 (c) 掺杂AB堆叠石墨烯的后处理合成常用程序的代表性温度分布，例如来自GO+固体或气体掺杂源的程序。 (d) 不同掺杂原子掺杂的闪蒸石墨烯的掺杂原子比和 (e) 质量产量。 (e) 不同掺杂原子掺杂的闪蒸石墨烯的掺杂原子比和 (e) 质量产量。

Figure 1. Direct synthesis of heteroatom-doped graphene by flash Joule heating. (a) Schematics of the synthesis of heteroatom-doped turbostratic graphene via flash Joule heating. (b) The real time temperature measurement during direct FJH synthesis of doped flash graphene. (c) Representative temperature profile of common procedures for the post-treatment synthesis of doped AB-stacked graphene, such as that from GO plus solid or gaseous heteroatom sources. (d) Heteroatom ratios and (e) mass yields for different heteroatom-doped flash graphene.

(图1c)。 延长反应时间导致形成杂原子掺杂AB堆叠石墨烯，其比涡轮层 (Figure 1c)。 The extended reaction time leads to the formation of heteroatom-doped AB-stacked graphene,¹ which is harder to exfoliate than turbostratic graphene.¹⁹

对于单一类型的掺杂，合成的FG可以达到高达7.4%的杂原子比，对于多种杂原子共掺杂，The as-synthesized FG can reach a heteroatom ratio up to 总杂原子比>10%，包括硼和氮(B, N-FG)以及硼、氮和硫(B, N, S-FG)，这表明7.4% for a single type dopant¹⁹ and a total heteroatom ratio >10% for multiple heteroatom codoping, including boron and nitrogen (B,N-FG) and boron, nitrogen, and sulfur (B,N,S-FG), demonstrating the broad applicability of the FJH process (Figure 1d).

FJH合成在厚壁聚合物管内以毫秒为单位进行，该厚壁聚合物管具有紧密贴合的石墨间隔物，以最小化掺杂剂的挥发。 This enables a high mass conversion reaction yield up to 74% (Figure 1e).

Characterization of Nitrogen-Doped Flash Graphene. 我们首先研究了氮掺杂石墨烯，因为它是一种有效的电催化剂。 We first explore the nitrogen-doped graphene, since it has been reported to be an effective electrocatalyst.^{19,20}

The incorporation of nitrogen into the honeycomb lattice of graphene can be diverse in its substitutional orientations, including pyridinic N, pyrrolic N, graphitic N, and nitrogen oxides (Figure 2a). These N-doped sites change the local symmetry along with electronic structure, and can behave as active sites for various electrochemical and electronic applications. Melamine (C₃H₆N₆) is our chosen dopant owing to its high nitrogen

67 wt%的比例。 合成的N-FG在N掺杂下具有良好的石墨烯质量。 ratio of ~67 wt %. The as-synthesized N-FG has good graphene quality with N doping. The average Raman spectrum is collected from 100 sampling points and the standard deviation is <5%, which reflects the high homogeneity of the N-FG (Figure 2b). High D peak (~1339 cm⁻¹, breathing mode), G peak (1570 cm⁻¹, 环中所有sp²碳原子的呼吸模式), G peak (1570 cm⁻¹, mode of sp²-carbon atoms in rings), G peak (~1570 cm⁻¹, 石墨中的二阶区域边界声子), 2D/G和D/G强度比分别为0.84和0.65, 表明存在丰富的N掺杂取代缺陷。 2D peak (~2672 cm⁻¹, second order zone boundary phonons in graphene)²⁴ can be observed simultaneously and the 2D/G and D/G intensity ratios are ~0.84 and ~0.65, respectively, indicating abundant N-doped substitutional defect sites.^{26,27}

In contrast, 固有FG具有低ID/IG<0.05和高ID_{2D}/IG~1.95。 高分辨率拉曼光谱显示存在TS₁(1873 cm⁻¹)和TS₂(2018 cm⁻¹)峰，并且没有M峰(1750 cm⁻¹) (图S4), ~1.95。 The high-resolution Raman spectrum shows the existence of N-FG的涡轮层定向堆叠。 The X-ray diffraction (XRD) results reflect a downshift of the (002) peak relative to parent

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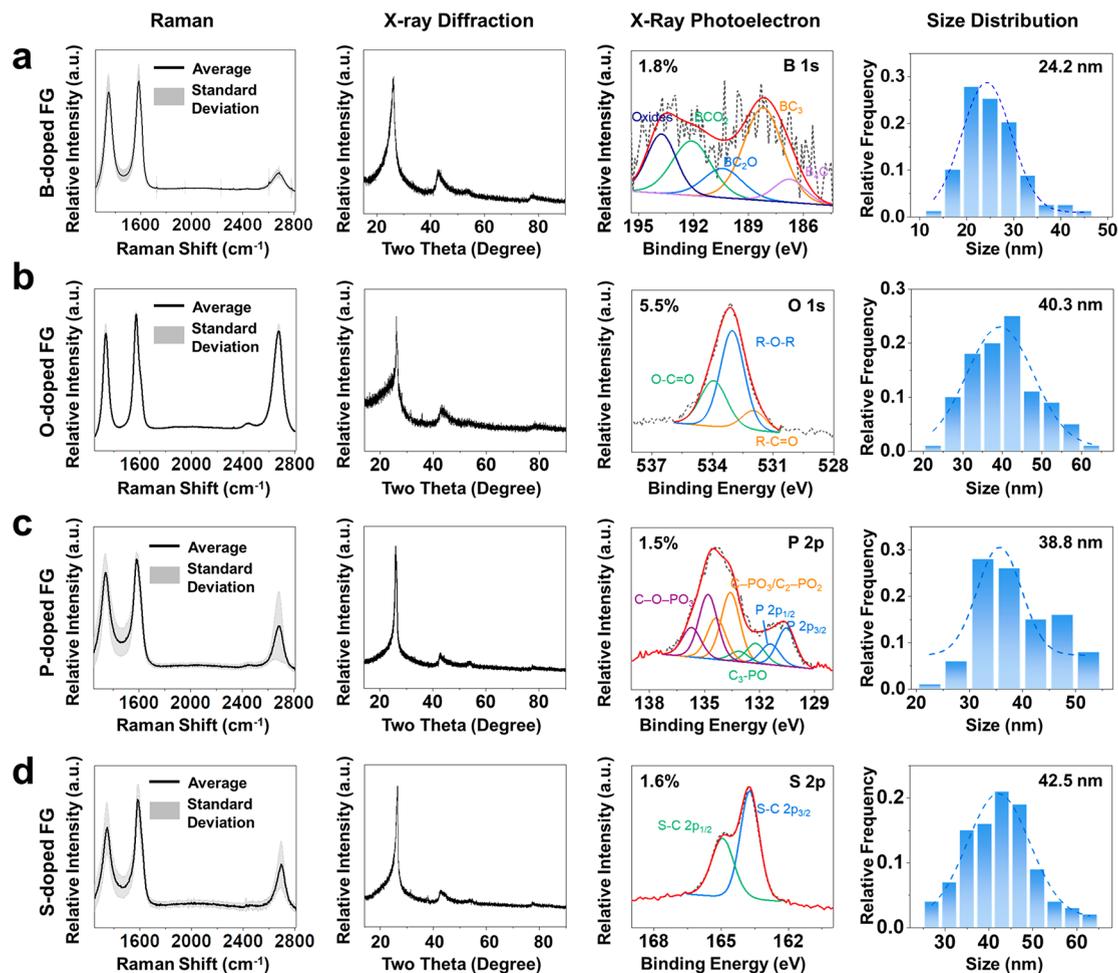


Figure 3. Characterization of heteroatom-doped flash graphene. Characterization results, including Raman spectra (showing the average and standard deviations over 100 points), XRD spectra, XPS results (showing the elemental ratios and the bonding states), and the size distributions of (a) boron-doped flash graphene, (b) oxygen-doped flash graphene, (c) phosphorus-doped flash graphene, and (d) sulfur-doped flash graphene. The statistical Raman results are collected from 100 sampling points. High-resolution P 2p spectrum shows the splitting of each peak to $2p_{1/2}$ and $2p_{3/2}$ portions, which are denoted by the same color. The SEM-derived sheet size distribution results are obtained from at least 100 graphene sheets.

粒径分布较窄(图2f),表明通过FJH合成可以精确控制石墨烯薄片的粒径,98.9 nm,且尺寸分布是窄的(Figure 2f), demonstrating the precise control of graphene flake size without obvious aggregation by FJH synthesis. The scanning electron microscopy (SEM) images show that N-FG is comprised of many nanocrystals (Figure 2g, Figure S9), and it is morphologically different from the undoped sheet-like FG, which usually has a lateral size up to several micrometers. Transmission electron microscopy (TEM) confirms the existence of nanoparticles and high-resolution TEM (HR-TEM) images show the presence of lattice fringes along with jagged structures (Figure 2h, Figure S10). Energy-dispersive spectroscopy (EDS) shows the existence of N-doping (Figure S11). Scanning TEM (STEM) and corresponding elemental mapping demonstrate the homogeneous distribution of the N doping by FJH synthesis (Figure 2i).

FJH Synthesis of Heteroatom-Doped Flash Graphene. The versatility of FJH synthesis is demonstrated and five distinct heteroatom-doped FG, including boron, nitrogen, oxygen, phosphorus, and sulfur-doped FG, are prepared (Figure 1a). The sealed polymer tubes prevent the dissipation of carbon and dopants, and a high mass conversion reaction yield (>70%) can be achieved as shown in Figure 1e. These heteroatoms have different atomic radii (50–100 pm)

compared to carbon (70 pm) (Figure S12). Therefore, there is some lattice mismatch in heteroatom-doped graphene, and the formation of substitutional solid solutions are limited to certain feasible ranges without the destruction of the graphitic structure. The electronegativities of these heteroatoms (2.0–4.0) are distinct from that of carbon (2.5), and thus the intrinsic electronic structure of graphene can be modified by different heteroatom dopings. Owing to the ultrahigh electrothermal temperature (>3000 K) during the FJH process, a wide variety of dopants, such as elements (M), oxides (MO_x), and organic compounds (C_xH_yM) can be used as precursors, regardless of their boiling points or conductivities. In contrast, post-treatment methods usually depend on the vapor pressure in the gas phase or the solubility in the solution phases. Thus, the FJH synthesis allows for the direct use of various solid dopants and enables the low-cost and mass production of heteroatom-doped graphene.

Boron forms bonding types similar to that of carbon, including graphitic B (BC_3), borinic B (C_2BO), and boronic B (CBO_2) patterns (Figure S12). The boron-doped FG (B-FG) has good graphitic structure with B doping. The average Raman spectrum indicates 2D/G and D/G intensity ratios are 0.26 and 0.91 (Figure 3a), which indicates the good graphene

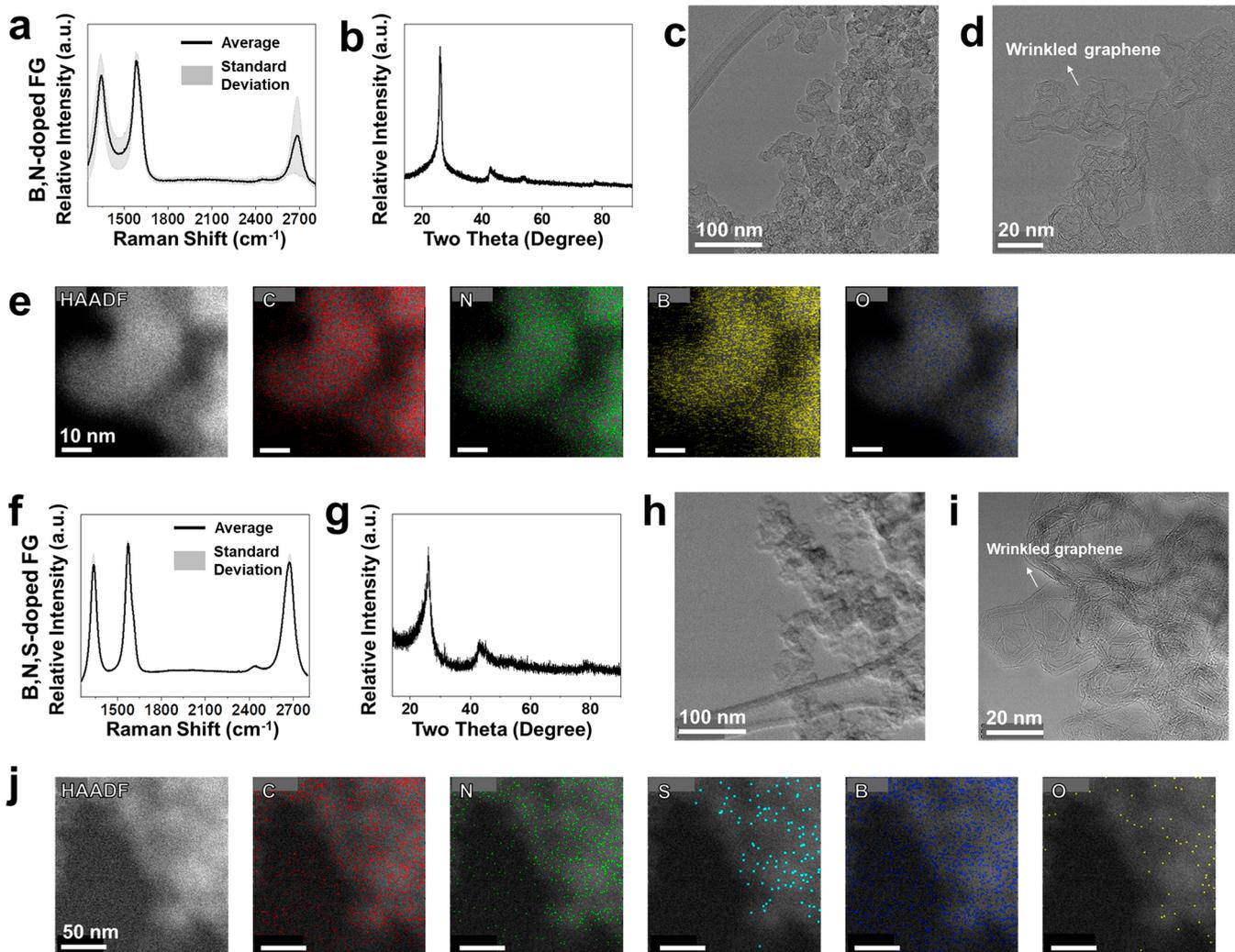


图4. 闪蒸石墨中多个共掺杂原子的表征。

Figure 4. Characterization of multiple heteroatoms codoped in flash graphene. (a) Raman spectrum of B, N codoped flash graphene. The black line and the gray shadow represent the average value and the standard deviation of 100 sampling points, respectively. The same applied to the Raman spectrum below. (b) XRD result of B, N codoped flash graphene. (c) TEM and (d) HR-TEM images of B, N codoped flash graphene. (e) STEM image and corresponding elemental distribution of B, N codoped flash graphene. (f) Raman spectrum of B, N, S codoped flash graphene. (g) XRD result of B, N, S codoped flash graphene. (h) TEM and (i) HR-TEM images of B, N, S codoped flash graphene. (j) STEM image and corresponding elemental distribution of B, N, S codoped flash graphene.

尽管同时引入了来自B掺杂的丰富缺陷。

quality despite the simultaneous introduction of rich defects derived from B doping. Since the average 2D/G intensity ratio and the statistical median of doped graphene is 0.10 (Table S3), a threshold value of 0.10 is used to determine whether the site of interest is graphitic. Therefore, the graphene yield of B-FG reaches 94% (Table S4). The interlayer spacing of the B-FG nanocrystals increases by 2.4% over that of commercial graphene (CG) and the average crystal size is 24.2 nm (Table S5 and Figure S14).

热力学相图表明, 当使用二十面体B₁₂作为掺杂剂并且硼的局部浓度达到>9%时, 其碳化物B₄C是最稳定的状态。另一方面, H₃BO₃前体有助于形成富硼的石墨结构, 而不会在XRD光谱中出现碳化物信号。B-FG中硼的原子浓度为1.8%。The atomic concentration of boron in B-FG is ~1.8%.

类似地, 其他气态元素(例如磷)和硫族元素(包括氧和硫)可以掺杂到FG中(图3b-d, 图S12)。Similarly, other pnictogens such as phosphorus, and chalcogens including oxygen and sulfur, can be doped into FG (Figure 3b–d, Figure S12). After heteroatom doping, the

石墨结构得到保留, 石墨含量产率为86–100%, 由拉曼映射确定(表S4), 而质量产率graphitic structure is preserved, and the graphene content yields are 86–100% determined by Raman mapping (Table S4), while the mass yields are shown in Figure 1e.

掺杂原子掺杂引入了大量的点缺陷, 降低了对称性。因此, 随后出现更高的D峰。与本征FG相比, 当ID/IG<0.05时, 这些掺杂FG的ID/IG更高, 为0.65–0.91(表S4)。Compared with intrinsic FG, with a low I_D/I_G < 0.05, these doped FGs have higher I_D/I_G of 0.65–0.91 (Table S4). The heteroatom-codoped FGs exhibit expanded interlayer distance up to 2.7% over that of CG, and likewise have larger surface area reaching ~876 m² g⁻¹ (Figure S15) over that of CG (~200 m² g⁻¹) and FG (~300 m² g⁻¹).

The doped FG nanocrystals have narrow flake size distributions and have abundant micropores as well as mesopores (Figure 3b–d, Figure S15–S16). The heteroatom contents are ~1.5% for P-FG, ~5.5% for O-FG, and ~1.6% for S-FG (Figure S17), which introduce VBM of -1.91 to -2.68 eV as shown in the valence band XPS results (Figure S18).

多原子共掺闪蒸石墨的表征。Dual doping and multiple doping can result in synergistic effects, which may lead to a stronger

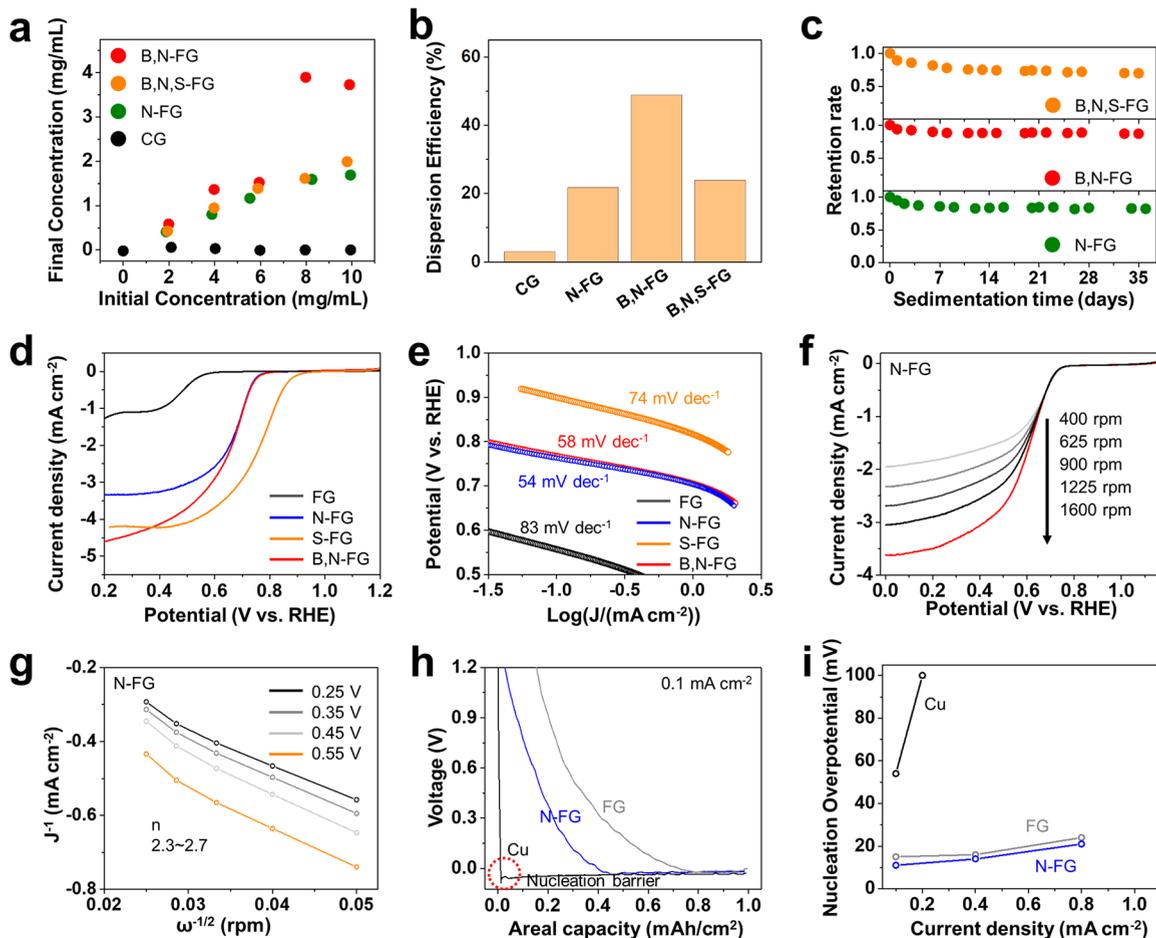


图5. 杂原子掺杂闪蒸石墨烯的应用。

Figure 5. Applications of heteroatom-doped flash graphene. (a) Various heteroatom-doped FG and commercial graphene (CG) dispersed in water/Pluronic (F-127) (1 wt %). (b) Dispersion efficiency of CG and heteroatom-doped FG. (c) The retention rate of different heteroatom-doped FG after recorded sedimentation times. (d) The ORR performance of different doped FG and intrinsic FG in 0.1 M KOH solution at 1600 rpm and (e) the corresponding Tafel plots. (f) The ORR polarization curves at different rotating rates of N-FG and (g) the corresponding K-L plots. (h) The voltage profile of Cu, FG, and N-FG during the Li metal electrodeposition. The testing condition is 0.1 mA cm⁻² with an areal capacity of 1.0 mAh cm⁻². (i) The nucleation overpotential of Cu, FG, and N-FG anodes in Li metal batteries at different current densities.

influence on charge density and spin density than a larger amount of single heteroatoms. These codoped heteroatoms with distinct chemical states can interact to provide active sites for enhancing electrocatalytic activity or introduce band structures along with local nanoscale effects. Likewise, mixed doping in more different species is higher when more distinct species are present, doping 2D materials with multiple types of heteroatoms is a viable strategy to obtain materials with high doping levels that would not be accessible with only a single dopant species.

In this section, B,N-FG and B,N,S-FG are synthesized by an ultrafast and all-solid-state catalyst-free FJH method. As-synthesized B,N-FG has a good graphene structure (Figure 4a), and the graphene content yield reaches 89% (Table S4). Dual doping causes abundant point defects and a high average $I_D/I_G \sim 0.90$ is obtained from 400 sampling points. The interlayer spacing of B,N-FG increases 2.6% (Figure 4b) compared to CG and the average particle size is 49.9 nm (Table S5, Figure S19). B,N-FG has a large surface area of 393 m² g⁻¹ and abundant micropores and mesopores (Figure S20). HR-TEM images show the presence of lattice fringes and wrinkled structures of doped FG nanoparticles. B, C, and N in these nanoparticles are uniformly distributed as shown in Figure 4e. High resolution XPS (Figure 4c,d). The uniform distribution of B, C, and N in these nanoparticles is shown in Figure 4e. High resolution XPS

spectra indicate that the total amount of B and N is 10.4% (Figure S21), which is higher than the fraction achieved when using only N or B dopant atoms alone. Emerging bonding states between heteroatoms, B and N, can be found in B 1s and N 1s spectra (Figure S21), and B,N-FG has a large VBM -3.02 eV (Figure S22).

Similar results are found in B,N,S-FG. The graphene content yield and I_D/I_G are 87% and 0.83, respectively (Figure 4f, Table S4). The interlayer spacing of B,N,S-FG is 2.2% larger than CG (Figure 4g) and the average particle size is 76.0 nm (Figure 4h, Figure S19). These nanocrystals exhibit similar wrinkled structures and homogeneous elemental distribution of B, N, and S (Figure 4i,j). The total atomic content for these three types of heteroatoms is $\sim 13.5\%$ and additional bonding states between heteroatoms, B and N, can be observed (Figure S23). The VBM of B,N,S-FG is -2.54 eV (Figure S22).

Scaling up and Applications of Heteroatom-Doped Flash Graphene. Gram-scale N-FG (N-FG-L2) synthesis is demonstrated in this section, and the graphene quality along with properties of N-FG-L2 are similar to those of N-FG synthesized from the small batch. The graphene content yield of 99% and 1.08 (Table S4, Figure S24). This reflects the point defects and hexagonal symmetric local structure of N-FG-L2 are 99% and 1.08, respectively (Table S4, Figure S24), which reflects the abundance of point defects

and local breaking of hexagonal symmetry. There is an asymmetrical (002) diffraction peak at ~ 26.0 degree, which indicates the expansion of interlayer distance by $\sim 2.1\%$. The N-FG-L2的平均尺寸为41.2 nm(图S25), 表面积为573 m²/g. N-FG-L2的孔径分布与N-FG相似。Pore size distribution of N-FG-L2 is similar to N-FG. The high-resolution N 1s spectrum shows the atomic ratio of N is $\sim 3.1\%$, and pyridinic N and pyrrolic N are dominant, which have strong binding energies with metals and other electropositive species.

杂原子掺杂的FG可分散在水-Pluronic(F-127)表面活性剂(1 wt%)溶液中, 与Heteroatom-doped FG is dispersible in water-Pluronic(F-127) surfactant (1 wt %) solution and similar to intrinsic FG, they give highly concentrated dispersions reaching ~ 4 mg mL⁻¹ (Figure 5a, Figure S26). 该结果与CG不同, CG在1 wt% F-127水溶液中的分散性很小(图5b, 图S27). CG, which shows little dispersibility in 1 wt % F-127 aqueous solution (Figure 5b, Figure S27). The dispersion efficiency measures the ratio between final dispersed concentration and initial concentration of graphene. The dispersion efficiencies of N-FG, B,N-FG and B,N,S-FG are ~ 8 , ~ 17 , and ~ 8 times higher than that of CG, respectively, which can be attributed to the turbostratic arrangement of heteroatom-doped FG allowing efficient exfoliation. 剥离所需的分散力远低于通过石墨剥离获得的传统AB堆叠石墨中的分散力。The dispersion force needed for exfoliation is much lower than that need in conventional AB-stacked graphene obtained by graphite exfoliation. The 分散液具有很高的时间稳定性, 35天后, N-FG、B、N-FG和B、N、S-FG的保留率分别为82%、86%和69%(图5c). 在其他有机溶剂中, 例如N、N-二甲基甲酰胺(DMF)和N-甲基-2-吡咯烷酮(NMP), N-FG也表现出良好的分散性。N-FG在DMF中的分散效率比CG高40.4倍和8.2倍(图S28).

pyrrolidone (NMP), N-FG also shows great dispersibility and the dispersion efficiencies of N-FG in DMF, and NMP solvents are ~ 40.4 and ~ 8.2 times higher than that of CG (Figure S28).³⁹ 为了证明杂原子掺杂FG作为电催化剂的适用性, 测试了各种掺杂FG的电化学ORR性能。To demonstrate the applicability of heteroatom-doped FG as electrocatalysts, the electrochemical ORR performances of various doped FG are tested. The polarization curves of five different doped FG and intrinsic FG are shown in Figure 5d and Figure S29. S-FG shows the most superior ORR activity among the metal-free electrocatalysts. The potential at 0.2 mA cm⁻² is 0.88 V and the Tafel slope is 74 mV dec⁻¹ (Figure 5e, Table S6). S-FG在1600 rpm时的饱和电流密度达到4.2 mA/cm². The saturated current density of S-FG at 1600 rpm reaches ~ 4.2 mA cm⁻². The electron transfer number (n) is important to determine the final product of ORR and it can be estimated by the Koutecký–Levich equation (K-L equation).⁴⁰ 不同转速下N-FG的极化曲线和相应的K-L图如图5f和图S30所示。The polarization curves of N-FG at different rotating rates and the corresponding K-L plots are shown in Figure 5f and g. And n is 2.3–2.7 for N-FG, which indicates H₂O₂ can be the major product. Heteroatom codoping results in distinct chemical states and synergistic effects. For B,N-FG, n is close to 4, which suggests H₂O is the major product (Figure S30). The stability tests of N-FG and B,N-FG show the increase of overpotentials and the decrease of current densities after 1000 cycles, which indicates the degradation of performance and might relate to the metastable structures (Figure S30). Since previous studies have shown that trace metals also contribute to the electrochemical performance of the carbon materials, we carried out ICP-MS to analyze the content of 14 different trace metals: Cr, Co, Fe, Mn, Ni, V, W, Pt, Mo, Ir, Pd, Au, Ti, and Ru, in the reactants and the products (Figures S31–34). The results show that the content of these metals from different samples are much lower than the values reported in the previous literature. 因此, 反应物和内参反应对产物的污染最小, 对ORR的影响很小。Therefore, the reactants and flash reaction will cause minimal contamination of the products and little effect on the ORR.

由于N中存在孤对电子, 其与金属原子(如Li)提供强大的结合能, 因此N-FG被用作裸金属电极的电极材料。Since there are lone pair electrons in N, which provides strong binding energies with metal atoms such as Li, N-FG is used as an electrode material in a lithium metal battery. 与裸铜电极(54 mV)相比, N-FG在0.1 mA/cm²下的金属锂沉积过程中表现出较小的形核过电位(11 mV)(图5h), 这表明N-FG的形核势垒较小。Compared to bare Cu electrode (54 mV), N-FG exhibits a smaller nucleation overpotential (11 mV) during the electro-deposition of metallic Li at 0.1 mA cm⁻² (Figure 5h), which indicates a smaller nucleation energy barrier for N-FG. A smaller energy barrier for nucleation facilitates the formation of more homogeneous nuclei and subsequent uniform metallic Li deposition. 较小的成核势垒有助于形成更均匀的原子核和随后均匀的金属锂沉积。The nucleation overpotential for N-FG is smaller than that of FG (Figure 5i), because the binding energy of N-FG is stronger than in intrinsic FG. This result is consistent with a previous literature report.

CONCLUSION

总之, 七种不同的杂原子掺杂FG, 包括单元素掺杂的N-FG、B-FG、O-FG、P-FG、S-FG、双元素共掺杂的B,N-FG和多元共掺杂的B,N,S-FG, 通过超快全固态无溶剂、无水和单元素掺杂的N-FG、B-FG、O-FG、P-FG、S-FG, dual elements codoped B,N-FG, and multiple elements codoped B,N,S-FG, are directly synthesized by an ultrafast and all-solid-state catalyst-, solvent-, and water-free FJH method. 无论沸点如何, 都可以使用不同的低成本掺杂剂, 包括单质、氧化物和有机化合物。Different low-cost dopants, including elements, oxides, and organic compounds can be used regardless of boiling point and conductivity. 合成的掺杂FG具有良好的石墨层质量、涡轮层状结构和扩展的层间距。The as-synthesized doped FG has good graphene quality, turbostratic structure, and expanded interlayer spacing. 因此, 它们可分散在水-Pluronic(F-127)(1 wt%)溶液中, 然后形成稳定的分散液。Therefore, they are dispersible in water-Pluronic(F-127)(1 wt %) solution, and then form stable concentrated dispersions. 杂原子掺杂改变了电子结构, 提高了掺杂FG作为电催化剂和电化学储能材料的性能。Heteroatom doping modifies the electronic structures, which improves the performance of doped FG as electrocatalysts and electrochemical energy storage materials. The gram-scale synthesis of doped FG is also demonstrated to show the scalability of the FJH method to bulk quantities. 杂原子掺杂FG合成的电能成本仅为1.2–10.7 kJ/g. The electrical energy cost for heteroatom-doped FG synthesis is only 1.2–10.7 kJ/g, which could render the FJH method suitable for low-cost and mass production of heteroatom-doped graphene. 由于目前使用类似的FJH技术将内在FG的产量扩大到每天1吨, 因此, 此处的技术易于转换为类似规模。Since the intrinsic FG is currently being scaled to 1-ton per day production using an analogous FJH technology,⁴⁶ the protocol here should be easily translated to similar scales.

CONCLUSION

实验部分

EXPERIMENTAL SECTION

材料。炭黑(CB, APS 10 nm, 黑珍珠2000)购自卡博特公司。Materials. Carbon black (CB, APS 10 nm Black Pearls 2000) was purchased from Cabot Corporation. Melamine (M2659-1KG, 99%) was purchased from Millipore-Sigma. Boric acid (31768-453.6G, 99.9%) 购自分析试剂。Boric acid (31768-453.6G, 99.9%) was purchased from Baker Analyzed Reagent. Sucrose (SX1075-1500g, ACS级) 购自Millipore-Sigma. (SX1075-1500g, ACS grade) was purchased from Millipore-Sigma. Red phosphorus (343242-5G, 99.99%) was purchased from 红磷(343242-5G, 99.99%) 购自管理博西格。Red phosphorus (343242-5G, 99.99%) was purchased from Alfa Aesar. Graphite spacers and tungsten carbide cylinder rods (4 mm diameter) were used as the electrodes. FJH反应容器采用聚四氟乙烯管、聚四氟乙烯或聚丙烯管(内径=4mm, 外径=25mm, 长度=6cm)。Alfa Aesar. Graphite spacers and tungsten carbide cylinder rods (4 mm diameter) were used as the electrodes. FJH reaction vessels were obtained from XG Sciences and were used as received for the comparison. Commercial graphene was obtained from Tianyuan Empire. 对于分散测试, Pluronic(F-127, P2443-250G)从Millipore Sigma购买。For the dispersion tests, the Pluronic (F-127, P2443-250G) was purchased from Millipore-Sigma. 对于电化学测试, 氢氧化钾(KOH), 221473-500G, ACS级) 从管理博西格公司购买。For the electrochemical tests, potassium hydroxide (KOH, 221473-500G, ACS grade) was purchased from Millipore-Sigma. 对于电池测试, N-甲基吡咯烷酮(NMP, 99.99%, 443778-500ML) 从管理博西格公司购买。For the battery tests, N-methyl pyrrolidone (NMP, 99.99%, 443778-500ML) was purchased from Millipore-Sigma. The milling ball (Yttrium stabilized ZrO₂, 99.5%, R = 5 + 0.3 mm) was purchased from MTI Corporation. Lithium bis(trifluoromethanesulfonyl)imide (LiTFSI, 99.95%, trace metals basis, 544094-25G) was purchased from Millipore-Sigma. Lithium

硝酸锂 (LiNO₃, 99.99%, 微量金属基, 229741-25G) 购自密理博西格玛。
 nitrate (LiNO₃, 99.99%, trace metals basis, 229741-25G) was purchased from Millipore-Sigma. 1,2-二甲氧基乙烷 (二甲醚, 无水, 99.5%
 1,2-dimethoxyethane (DME, 99.5% inhibitor-free, 259527-100ML) 购自密理博西格玛。
 anhydrous, 99.5% inhibitor-free, 259527-100ML) was purchased from Millipore-Sigma. 1,3-二氧六环 (DOL, 无水, 99.8%, 75 ppm BHT 作为抑制剂, 1,3-Dioxolane (DOL, anhydrous, 99.8%, ~75 ppm BHT) 购自 Millipore-Sigma. 1,3-Dioxolane (DOL, anhydrous, 99.8%, ~75 ppm BHT) was purchased from Millipore-Sigma. 聚偏氟乙烯粘合剂 (PVDF, 121120-80G) 购自 MTI 公司。
 Poly(vinylidene fluoride) binder (PVDF, 121120-80G) was purchased from MTI Corporation. 高导电乙炔炭黑 (ABHC-01342431) 购自 Soltex Corporation。
 High conductive acetylene black (ABHC-01, 342431) was purchased from Soltex Corporation. 对于电耦合等质子体质谱 (ICP-MS) 测试, HNO₃ (67.70 wt%, 痕量金属级, 赛希尔化学)。
 For the inductively coupled plasma mass spectrometry (ICP-MS) HCl (37 wt%, 99.99% 痕量金属基, 微孔西格玛), 水 (微孔西格玛, ACS 超痕量分析试剂), HNO₃ (67-70 wt %, TraceMetal grade, Fisher Chemical), HCl (37 wt %, 99.99% trace metals basis, Millipore-Sigma), water (Millipore-Sigma, ACS reagent for ultratrace analysis), and all the standards (periodic table mixtures) were purchased from Millipore-Sigma.

表征 使用 FEI Helios NanoLab 660 双光束扫描电镜, 在 5 kV 电压下, 工作距离为 4 mm, 通过扫描电镜 (SEM) 对反应物和内嵌产物进行了表征。
Characterization. The reactant and flash products were characterized through scanning electron microscopy (SEM) using a FEI Helios NanoLab 660 DualBeam SEM at 5 kV with a working distance of 4 mm. TEM images were taken with a JEOL 2100F field emission gun transmission electron microscope at 200 kV. HR-TEM and HAADF-STEM images were taken with FEI Titan Themis S/TEM instrument at 300 keV after accurate spherical aberration correction. X-ray photoelectron spectroscopy (XPS) data were collected with a PHI Quantera SXM Scanning X-ray Microprobe with a base pressure of 5×10^{-9} Torr. Survey spectra were recorded using 0.5 eV step sizes with a pass energy of 140 eV. Elemental spectra were recorded using 0.1 eV step sizes with a pass energy of 26 eV. All XPS spectra were corrected using the C 1s peaks (284.8 eV) as reference. X-ray diffraction (XRD) measurements were done by a Rigaku SmartLab Intelligent XRD system with filtered Cu K α radiation ($\lambda = 1.5406$ Å). Raman spectra were collected with a Renishaw Raman microscope using a 532 nm laser with a power of 5 mW. A 50 \times lens was used for local Raman spectra and Raman mapping. UV-vis (Shimadzu UV-3600 plus) was used to collect the spectra of the suspension of reactant and flash products. A Thermo Scientific Nicolet 6700 衰减全反射傅立叶变换红外光谱仪 (ATR-FTIR) 分析反应物和内嵌产物中的振动和旋转模式。
 The circuit diagram and FJH reaction box were shown in Figure S1. A thick-walled polymer tube (ID = 4 mm, OD = 25 mm, length = 6 cm) was prepared in a machine shop and used to load the reactant powder. Graphite spacers ($D = 4$ mm, length = 5 mm) and tungsten carbide cylinder rods ($D = 4$ mm, length = 5 cm) were utilized as electrodes to compress the loaded powder. The graphite spacers were in contact with the samples and tungsten carbide cylinder rods were connected with the circuit. In this work, different precursors were used to prepare heteroatom doped FG. The reactants included a mixture of (1) CB, (2) melamine and CB, (3) red phosphorus and CB, (4) sucrose and CB, (5) red phosphorus and CB, (6) PPS and CB, (7) boron, melamine, and CB, and (8) boron, melamine, PPS, and CB. The electrical energy was provided by a capacitor bank in the circuit with a total capacitance of 60 mF. The capacitor banks were charged by a d.c. supply that could reach 400 V. The flash duration was controlled by an Arduino controller relay in the circuit acting as a high-speed switch. The reactant resistances for various reactants were different, and the flash conditions were changed to control the total input electrothermal energy. To scale up N-FG-L1 (使用了厚壁聚合物管 (ID=8 mm, OD=25 mm, 长度=6 cm) 和石墨垫片 (D=8 mm, 长度=5 mm), thick-walled polymer tube (ID = 8 mm, OD = 25 mm, length = 6 cm) and graphite spacers ($D = 8$ mm, length = 5 mm) were used for N-FG-L1. The quartz tube (ID = 16 mm, OD = 20 mm, length = 6 cm) and graphite spacers ($D = 16$ mm, length = 3 cm) were used for N-FG-L2. And the input electrothermal energy was calculated based on the energy densities in Table S1. The parameters are listed in Table 1. The weight ratios of various samples are optimized in Table 1. There are four factors for which we try to optimize the final conditions:

1. Homogeneous flash reactions. We control the resistance of the evenly mixed reactants between 0.4–5.0 Ω , and monitor the

- 确保均匀闪蒸反应的电流-时间 ($I-t$) 曲线
 current-time ($I-t$) curves to ensure a homogeneous flash reaction
- The formation of turbostratic graphene. Raman and XRD are used to check the formation of turbostratic graphene to screen the condition.
 - The formation of bonds between sp² C and heteroatom. Since the purpose is to prepare doped graphene, the possibility of bonds between sp³ C and heteroatom should be excluded. XPS is used to analyze the bonding states.
 - The contents of the heteroatom. XPS is used to analyze the contents of the heteroatom.

有关 FJH 设置的更多安全说明和电路图, 请参阅支持信息和我们以前的出版物。
 More safety notes and a circuit diagram of the FJH setup are in the Supporting Information and our previous publications. **Caution!** 确保遵守所有安全预防措施以减轻触电。

Ensure that all safety precautions are observed to mitigate electrocution. As a last measure, be sure to wear thick rubber gloves that extend to the elbows. After the FJH reaction, it is suggested to allow the apparatus to cool and vent for 3–5 min. And ensure that the capacitor bank has been separated from the system by a circuit breaker (not a toggle switch), and that the capacitor bank has been fully discharged.

电化学测试 氧还原反应 (ORR)。
Electrochemical Tests. Oxygen reduction reaction (ORR). The slurry was prepared by dispersing 2 mg heteroatom doped FG in 2 mL water/ethanol (1:1, volume ratio). The slurry was sonicated for 30 min to form a homogeneous ink. A volume of 20 μ L of ink solution was loaded onto a glassy carbon electrode (5 mm in diameter) and dried in air at room temperature. The electrochemical measurements were carried out in a three-electrode configuration using a CHI 608D electrochemical workstation. A graphite rod and HgO/Hg (1 M KOH) electrode were used as the counter and reference electrodes, respectively. The electrolyte was 0.1 M KOH sparged with O₂ gas for O₂ saturation. CV was carried out with 0 V (vs RHE) of the potential range from 1.2 to 0 V (vs RHE) for 10 cycles at a scan rate of 100 mV s⁻¹. LSV was carried out with potential ranging from 1.2 to 0 V (vs RHE) at a scan rate of 5 mV s⁻¹.

For the battery tests, the electrode was prepared by grinding the mixture of doped FG, and poly(vinylidene fluoride) (PVDF) at a ratio of 0.9:0.1. A small amount (~3 \times of the total mass) of *N*-methylpyrrolidone was used to form a homogeneous slurry. The slurries were formed by ball milling at 1500 rpm for 10 min. The current collectors were Cu foil with a thickness of 10 μ m. The slurry was applied to the Cu foil by a doctor blade with a blade spacing of 100 μ m. The electrode was dried using a built-in heating cover placed on top of the electrode at 70 °C for 2 h and was then put in a vacuum oven overnight. The temperature and pressure of the vacuum oven were set at 70 °C and ~10 mmHg. The area of the cathode was ~2.0 cm². The electrolyte used was 1 M LiTFSI in a mixture of DME and DOL (V:V = 1:1) with 2% LiNO₃ as the additive. The volume of the electrolyte in each coin cell was 30 μ L.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsnano.2c01136>.

Safety instructions, TEM, SEM, and other images; XPS and Raman spectra; Additional graphs (PDF)

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Notes

The authors declare the following competing financial interest(s): Rice University owns intellectual property on flash graphene. Some of that intellectual property has been licensed to Universal Matter Inc. J.M.T. is a stockholder in Universal Matter, but he is not an officer, director, or employee. Conflicts of interest are mitigated through regular disclosures to and compliance with the Rice University Office of Sponsored Programs and Research Compliance. D.X.L. was employed by Rice University when this work was done. He is now employed by Universal Matter.

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