www.acsnano.org

# 杂原子掺杂闪蒸石墨烯 Heteroatom-Doped Flash Graphene

Weiyin Chen,<sup>#</sup> Chang Ge,<sup>#</sup> John Tianci Li, Jacob L. Beckham, Zhe Yuan, Kevin M. Wyss, Paul A. Advincula, Lucas Eddy, Carter Kittrell, Jinhang Chen, Duy Xuan Luong, Robert A. Carter, and James M. Tour\*



co-doped FG (boron and nitrogen), as well as three-element-co-doped FG (boron, nitrogen, and suffer a warlety of low-cost 平原、氧化物相對机行合物。 试Opanis, such as elements, oxides, and organic compounds are used. The graphene quality of heteroather HG, 福加印度间 and similar to intrinsic FG, the material exhibits turbostraticity, increased interlayer spacing, and superior dispersibility. 潤減」小同統原士掺涂片的电化字單並原反应,其甲硫掺杂FG的性能最好。 Electrochemical oxygen reduction of different heteroatom-doped FG is tested, and sulfur-doped FG shows the best 搜金属电泡测试表明,与铜或未掺杂FG相比,氦掺杂FG的成核过电位较小。 理金属电池测试表明, 与铜或未掺杂FG相比、氮掺杂FG的成核过电位较小。 performance. Lithium metal battery tests demonstrate that itrogen-doped FG exhibits a smaller nucleation overpotential 合成修杂原子的FG合成的电能成本仅为1.2到1.5% compared to Cu or undoped FG. The electrical energy cost for the synthesis of heteroatom-doped FG synthesis is only 1.2 to 10.7k/0\_1这可以使FIJ的方法适用于低成本大规模生产掺杂杂原子的不需做. I使FII方法适用于低成本大规模生产掺杂杂原子的石墨烯。 5, http://www.cost mass production of heteroatom-doped graphene. 10.7 kJ/g,这可以仍 10.7 kJ g , 关键词:闪蒸焦耳加热、直接合成、闪蒸石墨烯、杂原子掺杂、无催化剂 KEYWORDS: flash Joule heating, direct synthesis, flash graphene, heteroatom-doping, catalyst-free

# **INTRODUCTION**

本征纳米材料的化学功能化已广泛用于调整局部结构和电子状态,从而极大地影响材料的 Chemical functionalization of intrinsic nanomaterials has been widely used to tailor the local structures and electronic states, thereby greatly affecting the materials' optical, electrical, and 实现化学修饰的传统方法包括将各种杂原子、特 mechanical properties. Traditional methods to achieve mechanical properties. Traditional methods to achieve 定官能团、甚至其他分子引入固有晶格。 chemical modification involve the introduction of various heteroatoms,<sup>3,5</sup> specific functional groups,<sup>1,6</sup> or even other 空間子後沿是一种最基本但高效的改性 molecules into the intrinsic lattice. Heteroatom doping, which consists of substituting the original components with other elements, is the most basic yet highly effective modification.<sup>5,8</sup> 对于石墨烯,可以引入许多不同的掺杂剂,包括氮、硫、氧、硼、磷和一些金属原子。 For graphene, many different dopants can be introduced, including nitrogen, sulfur, oxygen, boron, phosphorus, and 杂扇子掺杂的石墨烯具有电催化性能,在传感应用中 some metal atoms. Heteroatom-doped graphene has 具有优越的性能,和与各种聚合物以及单个金属原子更强的相互作用。 demonstrated electrocatalytic properties, superior performance in sensing applications,<sup>5</sup> and stronger interactions with

various polymers as well as single metal atoms. 先前的研究主要集中在预制石墨烯晶格的优学功能化(后处理),包括球磨、氧化石 Previous research has focused on the chemical functionaliza-墨嘴(G)的热调火 警察子体外理和光化学后应 tion of preformed graphene lattice (post-treatment), including ball milling,<sup>5</sup> thermal annealing of graphene oxide (GO),<sup>13,14</sup> plasma treatment, and photochemical reactions. Ball 和等离子体法是合成杂原子掺杂石墨烯的简单且可扩展的方法,但掺杂通常局限于边缘或 milling and plasma methods are simple and scalable routes 世世知知 to synthesize heteroatom-doped graphene, but the doping is usually limited at the edges or other defects.<sup>5,15,16</sup> Thermal

热退火方法不仅需要GO作为在剧烈氧化下合成的前驱体,还需要延长高温以部分恢复sp2碳网络。 annealing method not only needs GO as the precursor, which is synthesized under harsh oxidation, but also requires extended high temperature to partially recover the sp<sup>2</sup> carbon 这些每上而下插入石墨烯通常会导致石墨烯质量的不可逆退化,这些步 network. These top-down insertions into graphene usually 哪可能会看问题 因为它们会导致过量在应反应的机构的CAMBCIOATOR的世界 network. These top-down insertions into graphicitic usual, 骤可能会有问题,因为它们会导致过量供应反应剂(如KMn04和BC3)不可避免地产 cause irreversible degradation of graphene quality, and these 生杂质。大多数自底向上直接合成掺杂杂原子的石墨烯可以通过化学气相沉积(CVD) procedures can be problematic since they result in inevitable 或合成有机策略实现,但产品仅限于超微量,从µg到mg,而且它们通常含有催化金属杂质 impurities from the excessive supply of reactive agents such as ,如路、紅相線。 KMnO<sub>4</sub> and BCl<sub>3</sub><sup>1,5</sup> Most bottom-up direct synthesis of heteroatom-doped graphene can be achieved by chemical vapor deposition (CVD)<sup>17</sup> or synthetic organic strategies,<sup>18</sup> but the products are limited to ultrasmall amounts, from  $\mu$ g- to mg-scales, and they often suffer from catalytic metal impurities 此外,可实现的异质掺杂石墨烯组合物的类型通常受到 such as Cr, Ti, and Ni. Furthermore, the types of heteroatom-前体蒸汽压或溶酶度的限制。 doped graphene compositions achievable are often restricted by the vapor pressure or solubility of the precursors.<sup>5</sup>

Received: February 2, 2022 Accepted: March 18, 2022



因此,从非晶碳源直接固相合成大量的异质掺杂石墨烯可以潜在地解决这些难题。 Therefore, the direct solid-phase synthesis of heteroatomdoped graphene in bulk quantities from amorphous carbon sources can potentially solve these dilemmas

sources can potentially solve these dilemmas. 在这里,我们揭示了一种在1秒内通过超快和全固态无催化剂快速焦耳加热(FJH)方 Here, we disclose a general synthetic route to heteroatom-法合成异质掺杂石墨烯的一般路线。 前体,在>10~5 K/s时将往面温度开周到>3500 K,然后在>10~4 K/s时快速冷却。 of high energy density is passed through the solid precursors, 七种重要的杂原子餐杂闪蒸石蜜烯(FG)、包括单元素掺杂(B,N,O,P,S)=元素  $^{-1}$ which raises the sample temperature to >3500 K at >10 K s 共掺杂(B,N)和三元素共掺杂(B,N,S)FG 通过最少闪蒸与各种低成本源(包 followed by rapid cooling at >10 K s . Seven types of 活元素 LetManAndCeny和备。 important heteroatom-doped flash graphene (FG), including single-element-doped (B, N, O, P, S), two-element-co-doped (B, N), and three-element-co-doped (B, N, S) FG are produced by millisecond flashing with various low-cost sources, including elements, oxides, and organic compounds. 这证明了利方法的多功能性。 正打該。demonstrates the versatility of the FJH method. These 些掺杂剂是因为它们可以提供空气稳定的掺杂石墨烯产品。 dopants were selected since they can afford air-stable doped 光谱分析表明,杂原子掺杂rG具有良好的石墨烯质量和修 graphene products. Spectroscopic analysis indicates that 協約电子结构。 heteroatom-doped FG has good graphene quality and modified 此外,我们证明了在纳米尺度上精确控制尺寸分布以及直接 electronic structure. Moreover, we demonstrate the precise 使用修杂原子的FG,而无需要力的纯化程序。 control of size distribution at the nanoscale and the direct usage of heteroatom-doped FG without laborious purification 研究了这些材料的分散性 并在电化学気还原反应(ORP 和理全層电 前先」」这些材料的分散性,并在电化学氧企原反应(ORR J和键金属电 procedures. The materials were studied for their dispersibility, and tested in electrochemical oxygen reduction reactions (ORR) and lithium metal battery (LMB) compositions. In addition, gram-scale synthesis of heteroatom-doped FG is explored to demonstrate the feasibility of scaling up its production.

#### 结果和讨论 RESULTS AND DISCUSSION

闪蒸焦耳加热直接合成掺杂杂原子的石墨烯。 Direct Synthesis of Heteroatom-Doped Graphene by Flash Joule Heating. In a typical flash synthesis process, the 「一初仕四丁配百民灯的口重空戶∠同的厚笙影百初官内做轻微比缩(图S1)。 powdery mixture of dopants and conductive carbon black is slightly compressed inside a thick-walled polymer tube between two well-fitted graphite spacers (Figure S1). The 聚合物管可以由聚四氟乙烯、聚(1.4-苯硫醚)或聚丙烯腈制成。 polymer tube can be made from Teffon, poly(1.4-phenylene 电路中的电容器组连接到石墨垫片,用于在小于1 sulfide), or polyacrylonitrile. The capacitor banks in the circuit are connected to graphite spacers and are used to provide electrothermal energy to the reactants in <1 s. Compared to 物中使用的松软铜棉电极相比,厚璧聚合物管和良好安装的石墨间隔大大减少了杂原子 fluffy copper wool-made electrodes used in our previous 的损失,并促进了染质子掺杂石量稀的形成。 publications, thick-walled polymer tubes and well-fitted graphite spacers greatly decrease the loss of heteroatoms and facilitate the formation of the heteroatom-doped graphene. 表1列出了详细的闪蒸参数。不同的掺杂剂可以用作前体,包括单质(M),氧化物( 夜19日〕 坪垣的内窓を致。 イロのりをホカリ 以用1F的座, 巴泊半原(101), 単いか) The detailed flash parameters are listed in Table 1. Different Mへい拓右和化全航(マHyM)(图12) dopants can be used as precursors, including elements (M), oxides (MO), and organic compounds (C, H, M) (Figure 1a). 在电压为10.V. 样品电阻为1.0的典型EII合成过程中,通过样常的电流在100m放电时 在中国的人類的人類的人類的研究的目的以及任何,通过性面的电流在100ms放电的 During a typical FJH synthesis process with a voltage of 110 V and a sample resistance of  $\sim 1.0 \Omega$ , the current passing through the sample reaches  $\sim 60$  A within  $\sim 100$  ms discharge time 以穆氮FG(N-FG)为例,总电能为6.8kJ(q(表S1), (Figure S2). Take nitrogen-doped FG (N-FG) as an example, the total amount of electrical energy is 6.6 kJ  $g^{-1}$  (Table S1). 通过拟合门口程中的黑体辐射光谱未测重温度(图33)。 The temperature is measured by fitting the blackbody radiation spectra during the FJH process (Figure S3). The transient x10^5 K/s时达到3500 K,随后55 ms后的超快冷却速率估计为1.8 x10^4 K/s(图 highest temperature reaches ~3500 K at ~2.0 × 10° K s<sup>-1</sup> and 1b, 这有助于形成具有较大层间间距的涡轮层结构。subsequent ultrafast cooling rate after ~55 ms is estimated as  $\sim 1.8 \times 10^4$  K s<sup>-1</sup> (Figure 1b), which facilitates the formation of turbostratic structures with larger interlayer spacings. This 快冷却不允许AB堆叠石墨烯发生,这有助于在溶剂中分散,如以前的出版物所示。 ultrafast cooling does not permit AB-stacked graphene to volatilizes the heteroatom components and enhances the homogeneous reactions between these precursor vapors and the amorphous carbon source to form the doped graphitic

#### 表1。不同系统的闪然参数 Table 1. Flash Parameters for Different Systems

	FG	N-FG	B-FG	O-FG
反应物 reactant	СВ	<mark>三聚氰胺+炭黑</mark> melamine + CB (75 wt %)	·····································	<del>蔗糖+炭黑</del> sucrose + CB (60 wt %)
样品质量 sample mass	30	40	40	40
(mg) 样品电阻 sample resistance	~0.7	~0.9	~2.0	~5.0
(Ω) 放电电压 discharge voltage (V)	110	110	110	120
闪蒸持续时间 flash duration	100	100	100	200
(ms) 闪蒸重复 flash	1 flash	1 flash	1 flash	3 flashes
repetitions 总电容 total	60	60	60	60
(mF) 腔室压力 chamber	Ar	Ar (~1 atm)	Ar (~1 atm)	Ar (~1 atm)

pressure (~1 atm)

	P-FG	S-FG	D,N-FG
反应物 reactant	<del>组读+炭黑</del> red phosphorus + CB (90 wt %)	<b>業年間+炭黒</b> PPS + CB (50 wt %)	#:=蹤瓢胺 boron:melamine:CB $(2:5:23)^a$
样品质量 sample mass (mg)	25	40	40
样品电阻 sample resistance	~1.5	~1.0	~3.0
(Ω) 放电电压 discharge voltage (V)	120	120	110
闪蒸持续时间 flash duration	200	100	50
(ms) 闪蒸重复 flash	3 flashes	1 flash	1 flash
repetitions 总电容 total	60	60	60
capacitance (mF) 腔室压力 chamber pressure	Ar (~1 atm)	Ar (~1 atm)	Ar (~1 atm)
DICOULLC			
1			
I	B,N,S-FG	N-FG	-L1 N-FG-L2
反应物 reactant	B,N,S-FG	S:CB melami	-L1 N-FG-L2 $me_{\pm}$ melamine + $mt \approx 10^{-10}$ CB (75 wt %)
反应物 reactant 样品质量 sample mass	B,N,S-FG	CB (75 320	-L1 N-FG-L2 me ± melamine + wt %) CB (75 wt %) 1000
反应物 reactant 样品质量 sample mass (mg) 样品电阻 sample resistance	B,N,S-FG B:=X400; X400; D boron:melamine:PPS (3:4:12:12) <sup>44</sup> 40 ~1.3	N-FG melam CB (75 320 ~1.0	-L1 N-FG-L2 melamine + wt %) CB (75 wt %) 1000 ~0.4
反应物 reactant       样品质量 sample mass (mg)       样品电用 sample resistance (Ω)       物电电压 discharge	B,N,S-FG ■:=XMB; X + MB; X boron:melamine:PPS (3:4:12:12) <sup>4</sup> 40 ~1.3 100	S:CB N-FG melami CB (75 320 ~1.0 155	-L1 N-FG-L2 melamine + melamine + CB (75 wt %) 1000 ~0.4 175
反应物 cactant       样品质量       sample       mass       (mg)       样品电阻       sample       resistance       (Ω)       放电电压       discharge       voltage (V)       闪蒸节线时间       flash duration	B,N,S-FG B:=X402; X402; Z boron:melamine:PPS (3:4:12:12) <sup>44</sup> 40 ~1.3 100 50	S:CB	-L1 N-FG-L2 melamine + wt %) CB (75 wt %) 1000 ~0.4 175 VFD <sup>b</sup>
反应物 reactant       样品质量 sample mass (mg)       #品电图 sample       resistance (Ω)       就电电压 discharge       voltage (V)       闪蒸持续时间 flash duration (ms)       闪蒸重复 flash	B,N,S-FG boron:melamine:PPS (3:4:12:12) <sup>44</sup> 40 ~1.3 100 50 1 flash	S:CB N-FG mglam CB (75 320 ~1.0 155 200 1 flash	-L1 N-FG-L2 melamine + wt %) CB (75 wt %) 1000 ~0.4 175 VFD <sup>b</sup> 1 flash
反应物 reactant       样品原量 sample mass (mg)       #品电阻 sample       resistance (Q)       が电电压 discharge       voltage (V)       闪蒸持续时间 flash duration (ms)       闪蒸重复 flash       repetitions       息电音	B,N,S-FG boron:melamine:PPS (3:4:12:12) <sup>44</sup> 40 ~1.3 100 50 1 flash 60	S:CB N-FG mglam CB (75 320 ~1.0 155 200 1 flash 222	-L1 N-FG-L2 melamine + wt %) CB (75 wt %) 1000 ~0.4 175 VFD <sup>b</sup> 1 flash 624
反应物 reactant         样品质量         sample mass (mg)         resistance         (mg)         按息电照         discharge         voltage (V)         风蒸持续时间         flash duration         (ms)         成素持续时间         flash         capacitance         (mF)         Boottage	B,N,S-FG boron:melamine:PPS (3:4:12:12) <sup>40</sup> 40 ~1.3 100 50 1 flash 60	S:CB N-FG mclam CB (75) 320 ~1.0 155 200 1 flash 222	$\begin{array}{c c} -L1 & N-FG-L2 \\ \hline melamine + \\ CB & (75 \text{ wt } \%) \\ \hline 1000 \\ \sim 0.4 \\ 175 \\ VFD^{b} \\ 1 \text{ flash} \\ 624 \end{array}$

"此处使用质量比"。 Mass ratios are used here. VFD指变頻驱动,是一种通过改变电源频率和持续时间来驱 动电气开关的控制器。 which is a type of controller that drives an electric switch by varying 在这里,使用10%占空 the frequencies and durations of its power supply. Here, 10% duty 比1s.然后使用20%占空比约。

因此,FJH工艺不需要使用额外的催化剂或过量的反应剂,从而避免了重复 structure. Therefore,FJH process does not require the use of 的后净化过程,如渗析。 extra catalysts or excessive reactive agents, which avoids redundant postpurification procedures, such as dialysis.<sup>12</sup> In 相反,传统的后处理掺杂方法需要(i)GO作为碳源,(ii)气相或液相杂原子源,以及( contrast, traditional post-treatment doping methods require (i) iii)长达12/时的反应时间 GO as the carbon source, (ii) gas- or solution-phase heteroatom sources, and (iii) long reaction time up to 12 h



图1、闪蒸焦耳加热直接合成掺杂杂原子的石墨烯。 Figure 1. Direct synthesis of heteroatom-doped graphene by flash Joule heating. (a) Schematics of the synthesis of heteroatom-doped turbostratic graphene via flash Joule heating. (b) 豫朶内蒸石墨烯直接FJH合成过程中的实时温度源量。 turbostratic graphene via flash Joule heating. (b) The real time temperature measurement during direct FJH synthesis of doped flash (c) [豫朶的堆叠石墨烯的后处理合成常用程序的代表性温度分布,例如年自GOH面体或气体不振子源的程序。 graphene. (c) Representative temperature profile of common procedures for the post-treatment synthesis of doped AB-stacked graphene, (d) 不同語 子校 (d) 不同語 (d) 不同語 (d) 不同語 (d) 不同語 (d) 不同語 (d) 不同意 (d) 不同語 (d) 不同語 (d) 不同意 such as that from GO plus solid or gaseous heteroatom sources. (d) Heteroatom ratios and (e) mass yields for different heteroatom-doped flash graphene.

白臺烯更准刻商。 formation of heteroatom-doped AB-stacked graphene,<sup>1</sup> which is harder to exfoliate than turbostratic graphene.<sup>19</sup> is harder \*minish 全成的IGOID认须到高达7.4%的杂原于比,对于多杂原子共掺杂 対于単一突型防漆が、台成時で回以応到高达(4%的深原手比、対于多保子狭窄が The as-synthesized FG can reach a heteroatom ratio up to 总杂原子比>10%、包括硼和氮(B, N-FG)以及硼、氮和硫(B, N, S-FG), 这表明 7.4% for a single type dopant and a total heteroatom ratio アに以工学的广泛法田姓(創石) 了时上之时》 泛迪用注( 智) 0 /s >10% for multiple heteroatom codoping, including boron and nitrogen (B,N-FG) and boron, nitrogen, and sulfur (B,N,S-FG), demonstrating the broad applicability of the FIH process HH合成在厚塑聚合物管内以電秒为单位进行,这厚塑聚合物管具有紧 (Figure 1d). The FJH synthesis takes place in milliseconds 密贴合的石墨间爆物,以最小化掺空和构建安 部局目的目室间隔初, 以取りいる家川和山本公。 within a thick-walled polymer tube with tightly fitting graphite spacers to minimize volatilization of the dopants. This enables a high mass conversion reaction yield up to 74% (Figure 1e). Characterization of Nitrogen-Doped Flash Graphene. 我们首先研究了氨掺杂石墨烯,因为它是一种有效的电催化剂。 We first explore the nitrogen-doped graphene, since it has been 氮在石墨烯蜂窝晶格 reported to be an effective electrocatalyst. The incorpo-中的掺入可以在其取代方向上不同。包括吡啶氮、吡咯氮、石墨氮和氨氧化物(图2a)。 ration of nitrogen into the honeycomb lattice of graphene can be diverse in its substitutional orientations, including pyridinic N, pyrrolic N, graphitic N, and nitrogen oxides (Figure 2a). 这些內掺杂位点随着电子结构的变化而改变局部对称性,并且可以作为各种电化学和电子应 These N-doped sites change the local symmetry along with electronic structure, and can behave as active sites for various electrochemical and electronic applications. Melamine  $(C_3H_6N_6)$  was our chosen dopant owing to its high nitrogen

67 wi%的比例。 ratio of ~67 wt %. The as-synthesized N-FG has good graphene quality with N doping. The average Raman spectrum deviation is <5%, which reflects the high homogeneity of the N-FG (Figure 2b). High D peak (339 cm-1, 环中配2線原子的呼 吸模式)、6鋒(1530 cm-1, 环中所有502碳原子对的键拉伸 1和20罐(2672 cm-1 mode of sp -carbon atoms in rings), G peak (~1570 cm<sup>-1</sup>, . 石墨烯中的一阶区域边界声子), 2D(5和D(6强度比分别为084和065, 表明存在丰富 MNi参全取代卦睑的, of all pairs of sp -carbon atoms in rings), and MNi参全取代卦睑的 2D peak ( $\sim 2672 \text{ cm}^{-1}$ , second order zone boundary phonons in graphene)<sup>24</sup> can be observed simultaneously and the 2D/Gand D/G intensity ratios are  $\sim 0.84$  and  $\sim 0.65$ , respectively, indicating abundant N-doped substitutional defect sites.<sup>26,27</sup> In 相反,固有FC具有低ID/IG<0.05和高IZD/IG-1.95。高分辨率拉曼光谱显示存在TS1( contrast,intrinsic FG has a low  $I_D/I_G < 0.05$  and a high  $I_{2D}/I_G$ 1873 cm-1)和52(2018 cm-1)锋,并且没有Multi (1750 cm-1)(图54), 1.95. The high-resolution Raman spectrum shows the 证实了N-FG的涡轮层定向准备。 existence of  $TS_1$  (~1873 cm<sup>-1</sup>) and  $TS_2$  (~2018 cm<sup>-1</sup>) peaks, and the absence of M peak (~1750  $\text{cm}^{-1}$ ) (Figure S4), substantiating the turbostratic oriented stacking of N-FG. 高分辨率俄歇光谱能够量化D参数,D参数测量微分CKLL光谱中最大值和最小值之间的能量 High-resolution Auger spectroscopy enables quantification of the D-parameter, which measures the energy separation between maxima and minima in differentiated C KLL N-FG的D參数为22.4 eV,接近石墨烯(图S5)。 spectrum. The D-parameter of N-FG is 22.4 eV and is close 太射线药制(XRD)结果反映了(002)嵴相对 to graphene (Figure S5). The X-ray diffraction (XRD) result 于研审社的工程 reflects a downshift of the (002) peak relative to parent



#### 石墨烯为25.9度,这表明层间间距扩大了2.7%(图2c)。 graphene to ~25.9 degree, which indicates the expansion of

无法观察到一般(100)和 interlayer spacing by ~2.7% (Figure 2c). The general (100) (101)衍射,并且可以看到(10)反射的不对称形状,其在高布拉格角侧具有明显的 and (101) reflections cannot be observed, and the asymmetric 地尾双辺。 shape of the (10) reflection, which has a pronounced tailing effect on the high Bragg angle side, can be seen. These results 面没有有序性叠 以及强轮层结构的存在。 国发有有序堆叠,以及涡轮层结构的存在。 support the absence of an ordered stacking of basal planes, and 由力硬源为炭黑, N-FG和 the existence of turbostratic structure. Since the carbon 其他後之商子的FC層干雅石墨化碟 其他掺杂杂原子的FG属于难石墨化碳。 source is carbon black, N-FG and the other heteroatom-doped 与易力量化碳( FG belong to the nongraphitizable carbons. Unlike the 日易 型 Unlike FG belong to the nongraphitizable carbons Unlike 也称为软碳) 不同,可石墨化碳在加热到3000 ° C时可以直接转化为晶体石墨,典型的 graphitizable carbon (also known as soft carbons), which could 進右團化低 / 他和為通知: / 在州2015年7天#和4447 4 为云墨 准有氧化碳( 也称为硬碱 )在电温度下小能浓转化为有辜。 be directly transformed into crystalline graphite when they were heated to the temperature up to 3000 °C, typical nongraphitizable carbons (also known as hard carbons) could not be thermally transformed into graphite at this temper-BL为中量是然刀字上最稳定的嫉形式,我们之前的研究发现,如果闪然持续 ature. Since graphite is the thermodynamically most stable 时间增加到几秒,石墨镭片的合并和AB排叠层的形成可能会发生。 form of carbon, our previous study found that the merging of graphene sheets and the formation of AB-stacked layers can happen if the flash duration is increased to several seconds.<sup>28</sup> N-FG的表面积为 $303 m^2/g$ , 丰富的孔径 $10 m(BS6)_{v^2} g^{-1}$  with abundant pore The surface area of N-FG is  $303 m^2 g^{-1}$  with abundant pore N-FG百两种类型的孔,封闭孔和开放孔。 N-FG有两种类型的孔,封闭孔和九放孔。 sizes <10 nm (Figure S6). N-FG has two types of the pores, 封闭和开始的法语取决于孔的位置和大小 closed pores, and open pores. The term of closed and open depends on the location and size of the pores into which a

测试流体(气体或液体)可能会扩散。 开放乱隙度和闭合孔隙度分 probe fluid (gas or liquid) can diffuse. The open porosity 别为0.54m0.1, and the closed porosity are 0.54 and 0.11, respectively.

新持杂曲X射线光电子能谱(XPS)证实。 N doping is confirmed by X-ray photoelectron spectroscopy 新行无案比为5.4%(图S7), 其中大多数以吡啶氮(208.7eV)和吡咯氮 (XPS). The elemental ratio of N is ~5.4% (Figure S7), the (399.8eV)的形式存在,没有氮氧化物的迹象(图2d,表52)。 majority of which is in the form of pyridinic N (~398.7 eV) and pyrrolic N (~399.8 eV), with no indication of N-oxides (Figure 2d, Table S2). The ratio of pyridinic N and graphitic N fakTyhon=fla@ag, 并与某些物种(如金属原子)表现出强烈的担意作用,这在电 reaches ~4.31. These pyridinic N atoms have unpaired 催化剂或金属电沉积中具有潜在的应用的信心。 electron spin density and exhibit strong interactions with some species, like metal atoms, which is useful in potential applications in electrocatalysts or metal electrodeposition. <sup>10,25</sup> measured for N-FG (Figure 2e). The N doping contents from 50%405.4%, 这证明了FH方法的可靠性。 3 different batches are ~5.5%, 5.0%, and ~5.4%, which N-FG的周期分子包括一个存在三聚氟酸酶以及CN和C N拉伸酶的信号,证实了 infrared (FTIR) spectra of N-FG show the absence of FHI过程中的氮掺杂(图8)。 melamine peaks and the signal of C—N and C=N stretching peaks in N-FG sample,<sup>5</sup> confirming the nitrogen doping during FJH process (Figure S8). <u>Rugicherer5程</u>, (02)的鲜的不对称展宽反映了N-FG的纳米级特征。 The asymmetric broadening of the (002) diffraction peak

The asymmetric broadening of the (002) diffraction peak reflects the nanosized feature of N-FG according to the 如个石墨烯片的平均尺寸为98.9 nm, Scherrer equation. The average size of 100 graphene sheets is



粒径分布较窄(图21),表明通过FJH合成可以精确控制石墨烯薄片的粒径, 98.9 nm and the size distribution is narrow (Figure 2f), 而没有明显的聚集。 demonstrating the precise control of graphene flake size without obvious aggregation by FJH synthesis. The scanning SEM)图像显示,N+G由许多纳米晶体组成(图2),图89),其形态不同于未掺杂片状 electron microscopy (SEM) images show that N-FG is FG,后者通常具有高达数微水的横向尺寸。 comprised of many nanocrystals (Figure 2g, Figure S9), and it is morphologically different from the undoped sheet-like FG, which usually has a lateral size up to several micrometers. <sup>19,28</sup> 遗射电子显微镜(TEM)证实了纳米颗粒的存在。高分辨率TEM(HR-TEM)图像显示存 Transmission electron microscopy (TEM) confirms the 在晶格条纹和锯齿结构(图2h,图510), existence of nanoparticles and high-resolution TEM (HR-TEM) images show the presence of lattice fringes along with figgged structures (Figure 2h, Figure S10). Energy-dispersive spectroscopy (EDS) shows the existence of N-doping (Figure S11). Scanning TEM (STEM) and corresponding elemental mapping demonstrate the homogeneous distribution of the N doping by FJH synthesis (Figure 2i).

FJH Synthesis of Heteroatom-Doped Flash Gra-证明 TiH合成的多功能性,并制备了五种不同的深原子掺杂FG,包括硼、氮 phene. The versatility of FJH synthesis is demonstrated and 氯、磷和硫掺杂FG(图1a)。 five distinct heteroatom-doped FG, including boron, nitrogen, oxygen, phosphorus, and sulfur-doped FG, are prepared 密封的聚合物管防止了碱和掺杂剂的喷溅,并且可以实现高质量转化反 (Figure 1a). The sealed polymer tubes prevent the dissipation 应产率(370%), 如图1e所示。 of carbon and dopants, and a high mass conversion reaction yield (>70%) can be achieved as shown in Figure 1e. These 与碳(70pm)相比,这些杂原子具有不同的原子半径(50-100pm)(图512)。 heteroatoms have different atomic radii (\$0-100 pm) Compared to carbon (70 pm) (Figure S12). Therefore, there 右墨衛中存在一些晶格失配,并且在不破坏石墨结构的情况下,取代固溶体的形成被限制在 is some lattice mismatch in heteroatom-doped graphene, and 一定的可行范围为。 the formation of substitutional solid solutions are limited to certain feasible ranges without the destruction of the graphitic structure. The electronegativities of these heteroatoms (2.0-(2.0-40)不同于镜(2.5), 因此可以通过不同的充原子指条水像彷石墨烯的固有电子结构。 4.0) are distinct from that of carbon (2.5), and thus the intrinsic electronic structure of graphene can be modified by different heteroatom dopings. Owing to the ultrahigh K), 各种各样的接杂剂,例如单质(M), 氧化物(MOX)和有机化合物(CMHW) electrothermal temperature (>3000 K) during the FJH process, 可以用作前体, 无论其漂点或电导率如何。 a wide variety of dopants, such as elements (M), oxides (MO<sub>x</sub>), and organic compounds (C<sub>x</sub>H<sub>y</sub>M) can be used as precursors, regardless of their boiling points or conductivities. 和反,后处理方法通常取决于气相中的蒸汽压或溶液相中的溶解度。 In contrast, post-treatment methods usually depend on the vapor pressure in the gas phase or the solubility in the solution mut, FIH合成化直接使用各种固体指示剂, 并且能够低成本和大规模生产掺 phases. Thus, the FJH synthesis allows for the direct use of 尔森原子的石墨礁。 various solid dopants and enables the low-cost and mass production of heteroatom-doped graphene. *硼*形成类似于碳的锂合类型,包括石墨的(CS)、硼酸B(CCB02) *细*+Directon forms bonding types similar to that of carbon,

構成である。 構成での「forms bonding types similar to that of carbon, 構成(図512)。 including graphitic B (BC<sub>3</sub>), borinic B (C<sub>2</sub>BO), and boronic B 接触FG(B-FG)負有良好的石墨结构。 (CBO<sub>2</sub>) patterns (Figure S12). The boron-doped FG (B-FG) 平均拉曼光谱表明 has good graphitic structure with B doping. The average 2D/G和D/G强度比分别为0.26和0.91(图3a), 这表明石墨烯质量良好, Raman spectrum indicates 2D/G and D/G intensity ratios are 0.26 and 0.91 (Figure 3a), which indicates the good graphene



图4。闪蒸石墨烯中多个共掺杂杂原子的表征。 Figure 4. Characterization of multiple heteroatoms codoped in flash graphene. (a) 疆氮共掺杂闪蒸石墨烯的拉曼光谱。 呼奖和古希伯斯公坦黑卡·በ尔平洋占约拉丁的在货店和法准查 marking mission with the gray shadow represent the average walke and the standard deviation of 100 sampling points, respectively. The same flash graphene. (c) (c) STEM (ingge and ing ing elemental distribution of B, N codoped flash graphene. (c) R (ingge and ingge elemental distribution of B, N codoped flash graphene. (c) R (ingge and ingge elemental distribution of B, N, S codoped flash graphene. (g) 就力能和例本任意的以外提到到结果。(h) 计 (h) 大作物的条件型的时间在通知的正确和(h) 计正确 image of B, N, S codoped flash graphene. (h) TEM and (i) HR-TEM images of B, N, S codoped flash graphene. (1) B.N. 3共管病闪然口臺海町十家川相应的元素方布。 (1) STEM image and corresponding elemental distribution of B, N, S codoped flash graphene.

尽管同时引入了来自B掺杂的丰富缺陷。 quality despite the simultaneous introduction of rich defects derived from B doping. Since the average 2D/G intensity ratio #且修宗石臺滿的统计中值为0,00(表33),因此使用0.000國道未确定感兴趣的位 of the conductive carbon reactant is 0.06 (Figure S13), and the 直走台入口室端。 statistic median of doped graphene is 0.10 (Table S3), a threshold value of 0.10 is used to determine whether the site of interest is graphene. 因此, B-FG的石墨烯产率达到94%(表84)。 interest is graphene. Therefore, the graphene yield of B-FG 与商用石墨烯(CG)相比, B-FG纳米晶体的层间距 reaches 94% (Table S4). The interlayer spacing of the B-FG 增加了24%, 平均晶粒尺寸为242nn(表85和图814)。 nanocrystals increases by 2.4% over that of commercial graphene (CG) and the average crystal size is 24.2 nm 热力学相图表明,当使用二十面体B12作为掺杂 (Table S5 and Figure S14). Thermodynamic phase diagrams 新進日期的目前效应注意。如此,此份能化物的公司是自己的状态。 剂并且硼的局部浓度达到>9%时,共价碳化物B4C是最稳定的状态。 indicate covalent carbide  $\mathrm{B_4C}$  is the most stable state when icosahedral B12 is used as the dopant and the local 第一方面, H3BO3詞体有期于形 concentration of boron reaches >9%. On the other hand, 成石墨结构,而不会在XRD光谱中出现碳化物信号(图3a)。 the  $H_3BO_3$  precursor facilitates the formation of graphitic structures without the appearance of a carbide signal in the XRD spectrum (Figure 3a). The atomic concentration of boron in B-FG is ~1.8%. 类似地, 其他气态元素(例如磷)和硫族元素(包括氧和硫)可以掺杂到FG中(图 Similarly, other pnictogens such as phosphorus, and

chalcogens including oxygen and sulfur, can be doped into FG (Figure 3b-d, Figure S12). After heteroatom doping, the

石墨结构得到保留,石墨烯含量产率为86~100%,由拉曼映射确定(表84),而质量产率 graphitic structure is preserved, and the graphene content yields are 86–100% determined by Raman mapping (Table S4), while the mass yields are shown in Figure 1e. The S4), while the mass yields are shown in AS-分原子掺杂引入了大量的点缺陷, 降低了对称性。 heteroatom doping introduces abundant point defects and beteroatom doping dopin have higher  $I_D/I_C$  of 0.65—0.91 (Table S4). The heteroatom-常比CG扩入72.7%,同样比CG(200 m2/g)和FG(300 m2/g)的表面和更大, doped FGs exhibit expanded interlayer distance up to 2.7% 法期政務m2/g(18515) over that of CG, and likewise have larger surface area reaching  $\sim 876 \text{ m}^2 \text{ g}^{-1}$  (Figure S15) over that of CG ( $\sim 200 \text{ m}^2 \text{ g}^{-1}$ ) and FG ( $\sim 300 \text{ m}^2 \text{ g}^{-1}$ ). The doped FG nanocrystals have narrow 微孔和中孔(图3bd, 835586), flake size distributions and have abundant micropores as well. as mesopores (Figure 3b-d, Figure S15-S16). The heter-P-FG的1.5%、O-FG的5.5%和S-FG的1.6%(图S17),其引入1.91至2.68eV的VBM,如价带 oatom contents are ~1.5% for P-FG, ~5.5% for O-FG, and ~1.6% for S-FG (Figure S17), which introduce VBM of -1.91to -2.68 eV as shown in the valence band XPS results (Figure <u>S18).</u>

8)。 多杂原子共掺闪蒸石墨烯的表征 Characterization of Multiple Heteroatoms Codoped 亚络九和名法公司以产生协同效应,这可能比大量 in Flash Graphene. Dual doping and multiple doping can result in synergetic effects, which may lead to a stronger



图5. 杂原子掺杂闪蒸石墨烯的应用。 Figure 5. Applications of heteroatom-doped flash graphene. (a) 分散在水/Pluronic(F-127)(1 wt%) 中的各种杂质子掺杂FG和商用石墨烯(CG)。 Figure 5. Applications of heteroatom-doped flash graphene. (a) Various heteroatom-doped FG and commercial graphene (CG) dispersed in (b) C3和杂质子掺杂FG的卷数率。 water/Pluronic (F-127) (1 wt %). (b) Dispersion efficiency of CG and heteroatom-doped FG. (c) The retention rate of different heteroatom-(d) Tale&PG and theteroatom-doped FG. (c) The retention rate of different heteroatom-(d) Tale&PG and theteroatom-doped FG. (c) The retention rate of different heteroatom-(d) Tale&PG and theteroatom-doped FG. (c) The retention rate of different heteroatom-(d) Tale&PG and theteroatom-(d) Tale&PG and theteroatom (ORR) performance of different doped FG and intrinsic FG in (f) N+GATC和特选下的ORR版化曲线和(g) 和应的化和。 (g) the corresponding K-L plots. (h) The voltage profile of Cu, FG, and N-FG during the Li metal electrodeposition. The testing mA/cm2, 面积容量为10mAffcm2, -2 with an areal capacity of 1.0 mAh cm<sup>-2</sup>. (i) The nucleation overpotential of Cu, FG, and N-FG anodes in Li metal batteries at different current densities.

光谱表明,B和N的总量为10.4%(图S21),高于仅使用N或B掺杂原子时获得的分数。 spectra indicate that the total amount of B and N is 10.4% (Figure S21), which is higher than the fraction achieved when 在B15和N16光谱中可以 发现究原子和N之间出现的键合状态(图S21),并且B,N-FG具有较大的VBM302eV( states between heteroatoms, B and N, can be found in B 1s and 图S22)。 N 1s spectra (Figure S21), and B,N-FG has a large VBM -3.02 eV (Figure S22).

eV (Figure S22). AB, N S-FGP+0发现了类似的结果, Similar results are found in B,N,S-FG. The graphene content  $87\%400.83(\mathbb{B}^4, \frac{8}{28})$ , yield and  $I_D/I_G$  are 87% and 0.83, respectively (Figure 4f, Table S4). The interlayer spacing of B,N,S-FG is 2.2% larger  $\mathbb{B}50$ , CG (Figure 4g) and the average particle size is 76.0 nm (Figure 4h, Figure S19). These nanocrystals exhibit similar  $37\pi_S^2\gamma^4n(\mathbb{B}^4, j)$ . The total atomic content for these  $\mathbb{E}R^{7}610\lambda(263)$ ,  $\mathbb{E}R^{3}$ 

**Scaling up and Applications of Heteroatom-Doped** 本节演示了克茲N+FG(N+FG-L2)合成, N+FG-L2的石墨烯质 **Flash Graphene**. Gram-scale N-FG (N-FG-L2) synthesis is 量和性能与小批量合成的N-FG相似。 demonstrated in this section, and the graphene quality along with properties of N-FG-L2 are similar to those of N-FG N-FG-L2的石墨烯含量产率和D/G分别 synthesized from the small batch. The graphene content yield 为999和(Nol 表 4, 图524), 这反映了点缺陷和六角形对称的局部破缺的丰富性. and  $I_D/I_G$  of N-FG-L2 are 99% and 1.08, respectively (Table S4, Figure S24), which reflects the abundance of point defects and local breaking of hexagonal symmetry. There is an  $\frac{626.0 \text{g} \text{b} \text{g}^{-7}}{\text{Tybrb}}$  asymmetrical (002) diffraction peak at ~26.0 degree, which indicates the expansion of interlayer distance by ~2.1%. The Nr-6-L20 multiple to the expansion of interlayer distance by ~2.1%. The Nr-6-L20 multiple to the expansion of interlayer distance by ~2.1%. The Nr-6-L20 multiple to the full standard state of N-FG-L2 is 41.2 nm (Figure S25) and the full surface area is 573 m<sup>2</sup> g<sup>-1</sup>. Pore size distribution of N-FG-L2 is similar to N-FG. The high-resolution N Is spectrum shows the  $\pm \frac{1}{2} \frac{1}$ 

they give highly concentrated dispersions reaching ~4 mg mL<sup>-1</sup>(图a, 图526)。 (Figure 5a, Figure 526). This result is distinct from 液中的分散性微小(图5b, 图527)。 CG, which shows little dispersibility in 1 wt % F-127 aqueous 分散效率衡量石墨烯的最终分散浓度 solution (Figure 5b, Figure S27). The dispersion efficiency 和初始浓度之间的比率。 和初始浓度之间的几乎。 measures the ratio between final dispersed concentration and initial concentration of graphene. The dispersion efficiencies of 利是CG的8、1/和8倍,这可以归因于杂原子掺杂FG的涡轮层排列允许有效刺离。 N-FG, B,N-FG and B,N,S-FG are ~8, ~17, and ~8 times higher than that of CG, respectively, which can be attributed to the turbostratic arrangement of heteroatom-doped FG 剥离所需的分散力远低于通过石墨剥离获得的 allowing efficient exfoliation. The dispersion force needed for exfoliation is much lower than that need in conventional AB-stacked graphene obtained by graphite exfoliation. The 分散波見有視亮的时间稳定性 35天后 N-FG B N-FG和B N S-FG的保留率分别为82% 力取放其有很高的时间稳定性,35天后,N-FG、b、N-FG和B、N、5-FG的休留学力加入2% dispersions show high temporal stability, and after 35 days, the retention rates of N-FG, B,N-FG and B,N,S-FG are 82%, 86%, and 69%, respectively (Figure 5c). In other organic solvents, 甲酰胺(DMF)和N-甲基-2-吡咯烷酮(NMP), N-FG也表现出良好的分散性, N-FG在 such as N,N-dimethylformamide (DMF) and N-methyl-2pyrrolidone (NMP), N-FG also shows great dispersibility and the dispersion efficiencies of N-FG in DMF, and NMP solvents are ~40.4 and ~8.2 times higher than that of CG (Figure

S28)<sup>39</sup> 为了证明杂原子掺杂FG作为电催化剂的适用性,测试了各种掺杂FG的电化学ORR性能。 To demonstrate the applicability of heteroatom-doped FG as electrocatalysts, the electrochemical ORR performances of various doped FG are tested. The polarization curves of five different doped FG and intrinsic FG are shown in Figure 5d and Figure S29. S-FG shows the most superior ORR activity among the metal-free electrocatalysts. The potential at 0.2 mA 增非沉斜率为/4mV/dec(156, 386)。 cm is 0.88 V and the Tafel slope is 74 mV dec<sup>-1</sup> (Figure 5e, Table S6). The saturated current density of S-FG at 1600 rpm reaches ~4.2 mA cm<sup>-2</sup> 由子转移数(n)对于确定ORR的最终产物很重要, 可以通过库特基列维奇方程(K-L方程)进行估计。 important to determine the final product of ORR and it can be estimated by the Koutecký-Levich equation (K-L equation).40 个同转速下N-FG的极化曲线和相应的K-L图如图5种的所示。 The polarization curves of N-FG at different rotating rates and the corresponding K-L plots are shown in Figure 5f and g. And n is 2.3-2.7 for N-FG, which indicates H<sub>2</sub>O<sub>2</sub> can be the major 京原于共修守政不同的代子私总和协同效应。 product. Heteroatom codoping results in distinct chemical states and synergistic effects. For B,N-FG, *n* is close to 4, which N-rられ時,N-rGHJ suggests H2O is the major product (Figure S30). The stability 稳定性测试表明,在1000次循环后,过电位增加,电流密度降低,这表明性能下降,可能 tests of N-FG and B,N-FG show the increase of overpotentials and the decrease of current densities after 1000 cycles, which indicates the degradation of performance and might relate to the metastable structures (Figure S30). Since previous studies 也有助于城材料的电化学性能,找们进行了ICP-MS,以分析反应物和产物中14种 have shown that trace metals can also contribute to the 不同微量金属的含量:Cr, Co, Fe, Mn, Ni, V, W, Pt, Mo, Ir, Pd, Au<sub>44</sub>, 所和KH 图 electrochemical performance of the carbon materials, we carried out ICP-MS to analyze the content of 14 different trace metals: Cr, Co, Fe, Mn, Ni, V, W, Pt, Mo, Ir, Pd, Au, Ti, and Ru, in the reactants and the products (Figures S31-34). The 结果表明,不同样品屮这些金属的含重远低于先则又厭甲报告的值。 results show that the content of these metals from different samples are much lower than the values reported in the 困此<sub>43</sub>反应物和闪蒸反应对产物的污染最小,对ORR的影 previous literature. Therefore, the reactants and flash reaction will cause minimal contamination of the products and little effect on the ORR.

由于N中存在孤对电子,其与金属原子(如Li)提供强大的结合能,因此N-FG被用作 Since there are lone pair electrons in N, which provides 理金重电池的电极材料。 strong binding energies with metal atoms such as Li, N-FG is used as an electrode material in a lithium metal battery.<sup>10</sup> 与裸铜电极(54mV)相比, N-FG在0.1mA/cm2下的金属锂电沉积过程中表现出较小 Compared to bare Cu electrode (54 mV), N-FG exhibits a 的形核过电位(11mV)(图h),这表明N-FG的形核刻垒较小, smaller nucleation overpotential (11 mV) during the electrodeposition of metallic Li at 0.1 mA cm<sup>-2</sup> (Figure 5h), which indicates a smaller nucleation energy barrier for N-FG. A 较小的成核势垒有助于形成更均匀的原子核和随后均匀的金属锉沉积, smaller energy barrier for nucleation facilitates the formation of more homogeneous nuclei and subsequent uniform metallic Li MCFON成核过电位小于FG(图5i), 因为N-FG的经能力和FG ge\_sition. The nucleation overpotential for N-FG is 要强. smaller than that of FG (Figure 5i), because the binding energy of N-FG is stronger than in intrinsic FG. This result is atMather a previous literature report.<sup>10</sup>

#### CONCLUSION

总之,七种不同的杂原子掺杂FG,包括单元素掺杂的N-FG,B-FG、O-FG、P-FG、S-FG、双 In conclusion, seven different heteroatom-doped FG, including 元素共進名的N.N-FG和多元要共進金的N.N-FG和多元要認知者の目的主要が知道。 元素共掺杂的B.N-FG和多元素共掺杂的B.N.S-FG,通过超快全固态无催化剂、无溶剂和无水 single element doped N-FG, B-FG, O-FG, P-FG, S-FG, dual Fill名注着接合成 elements codoped B,N-FG, and multiple elements codoped B,N,S-FG, are directly synthesized by an ultrafast and all-solid-大论滞点和 state catalyst-, solvent-, and water-free FJH method. Different 电导率如何,都可以使用不同的低成本掺杂剂,包括单质、氧化物和有机化合物。 low-cost dopants, including elements, oxides, and organic compounds can be used regardless of boiling point and 合成的接杂FG具有良好的石墨像质量 温纶层状结构和扩展的层间距。 言成的修宗FG具有良好的有靈婦质重、涡轮层状结构和扩展的层间距。 conductivity. The as-synthesized doped FG has good graphene quality, turbostratic structure, and expanded interlayer spacing. 因此,它们可分散在水Pluronic(F-127)(1)\t%)溶液中,然后形成稳定的浓缩分 Therefore, they are dispersible in water-Pluronic (F-127) (1 wt 散逸。 %) solution, and then form stable concentrated dispersions. 杂原子掺杂改变了电子结构,提高了掺杂FG作为电催化剂和电化学储能材料的性能。 Heteroatom doping modifies the electronic structures, which improves the performance of doped FG as electrocatalysts and electrochemical energy storage materials. The gram-scale 明了时用方法对生产的可扩展性。 synthesis of doped FG is also demonstrated to show the 余原子疹余⊦G合成 scalability of the FJH method to bulk quantities. The electrical <sup>107电能成争(X,7)1,2-10,7 k/g,</sup> energy cost for heteroatom-doped FG synthesis is only 1.2-这可以使け后法适用于低成本和大规模生产掺杂杂原子的石墨筛。 10.7 kJ g , which could render the FJH method suitable for low-cost and mass production of heteroatom-doped graphene. 由于目前使用类似的FJH技术将内在FG的产量扩大到每天1吨,因此,此处的技术应易于 Since the intrinsic FG is currently being scaled to 1-ton per day production using an analogous FJH technology,<sup>46</sup> the protocol here should be easily translated to similar scales.

#### 实验部分 EXPERIMENTAL SECTION

硝酸锂(LiNO3,99.99%,微量金属基,229741-25G)购自密理博西格码。 nitrate (LiNO<sub>3</sub>, 99.99%, trace metals basis, 229741-25G) was 12℃一面質素尤能(℃用量,无水,99.5%) 12-二甲氧基乙烷(二甲種), 无水, 95% 12-二甲氧基乙烷(二甲種), 无水, 95% 无抑制剂, 259527-100ML)购自密理博西格玛。 anhydrous, 99.5% inhibitor-free, 259527-100ML) was purchased 13-二氧戊环(DOL, 无水, 93%, 75pm BHT作为抑制剂, from Millipore-Sigma. 1,3-Dioxolane (DOL, anhydrous, 99.8%, ~75 27(100-1), 地區戶Millipore) 271020-1L)例目Milliporesigna, ppm BHT as inhibitor, 271020-1L) was purchased from Millipore-聚催氧乙醛联合剂(PVDE 121120,80G)除自MTI公司 家頃報乙烯稻音剂(PVDF, 121120-80G)胸目MIT经月。 Sigma. Poly(vinylidene fluoride) binder (PVDF, 121120-80G) was purchased from MTI Corporation. High conductive acetylene black

Soltex Comportion, (ABHC-01, 342431) was purchased from Soltex Corporation, 対于电應耦合等离子体质谱(ICP-MS)测试, HNO3(6770 wt%, 痕量金属级, 费希尔化学) For the inductively coupled plasma mass spectrometry (ICP-MS) HC(37 wt, 993%福金麗華, 滑力路挡, 次(滑力高相率, ACS銀麗愛分析试剂) tests, HNO3, (67-70 wt %, TraceMetal grade, Fisher Chemical), HCl 利所有标准品/周期表信念物/均从微力面格玛购买。 (37 wt %, 99.99% trace metals basis, Millipore-Sigma), water (Millipore-Sigma, ACS reagent for ultratrace analysis), and all the standards (periodic table mixtures) were purchased from Millipore-Sigma.

ma. 養証 使用FEI Helios NanoLab 660双光束扫描电镜,在5kV电压 **Characterization**. The reactant and flash products were 工作距离为4mm,通过扫描电子显微镜(SEM)对反应物知闪差字物进行了赛征。 下,工作距离为4mm,通过扫描电子显微镜(SEM)为反应初和内然广初进行了表现。 characterized through scanning electron microscopy (SEM) using a FEI Helios NanoLab 660 DualBeam SEM at 5 kV with a working 使用 EOL 2100F场发射抢透射电子显微镜在200 kV下拍摄TEM图像。 distance of 4 mm. TEM images were taken with a JEOL 2100F field 在時期的時度 emission gun transmission electron microscope at 200 kV, HR-TEM 校正后、使用FLITtan Themis \$77EM仪器在300 keV下拍摄HR-TEM和HAADE-\$TEM图像。 校正后 , 使用EI Titan Themis S/TEM仪器在300 keV下拍摄HR-TEM和HAADF-STEM图像。 and HAADF-STEM images were taken with FEI Titan Themis S/ TEM instrument at 300 keV after accurate spherical aberration 使用PHI-Quantera SXM扫描X射线微探针在5×109托的基压下收集X射线光电子 correction. X-ray photoelectron spectroscopy (XPS) data were 能電(XPS)数据。 collected with a PHI Quantera SXM Scanning X-ray Microprobe 使用0.5AV步长记录测量光谱、通过能量 with a base pressure of  $5 \times 10^{-9}$  Torr. Survey spectra were recorded 度用U.10V少でなる using 0.5 eV step sizes with a pass energy of 140 eV. Elemental spectra 変光谱、通过能量为26.0V. 素光谱,通过能量为26.6V。 were recorded using 0.1 eV step sizes with a pass energy of 26 eV. All 使用C1編(284.6V)作为参判所有XP3光谱进行校正。 of the XPS spectra were corrected using the C 1s peaks (284.8 eV) as X的技術到(XRD)测量相kigak/SmartLab置能RD系统进行,过滤铜钾辐射 reference. X-ray diffraction (XRD) measurements were done by a (=15406). ( =1.5406)。 Rigaku SmartLab Intelligent XRD system with filtered Cu K $\alpha$ 使用功率为5 mW的532 nm激光,使用Renishaw拉曼显微 radiation ( $\lambda = 1.5406$  Å). Raman spectra were collected with a 能比集功率光谱 i酸u集拉曼光谱、
 Renishaw Raman microscope using a 532 nm laser with a power of 5 (中用0x 透镜进行局部拉曼光谱和拉曼映射。
 mW. A 50× lens was used for local Raman spectra and Raman (中用U+vis (急建U+3000 plus)) 地质反应物和闪蒸产物悬浮液的光谱。
 mapping. UV-vis (Shimadzu UV-3600 plus) was used to collect the (中用Uhermo) 使用Thermo genetific Nicolet 6700衰減全反射傅立叶变换红外光谱仪(ATR-FTR)分析反应物和闪蒸产 Scientific Nicolet 6700衰減全反射傅立叶变换红外光谱仪(ATR-FTR)分析反应物和闪蒸产 Scientific Nicolet 6700 attenuated total reflectance Fourier-transform infrared (ATR-FTIR) spectrometer (Waltham, MA) was used to analyze the vibration and rotation modes in the reactant and the flash product.

EIH系统 电路图和FJH反应箱如图SI所示。 FJH System. The circuit diagram and FJH reaction box were 在現著年间創畫厚壁聚合物管(ID=4mm, OD=25mm, 长度=6cm), shown in Figure S1. A thick-walled polymer tube (ID = 4 mm, OD = 并用于装载反应物粉末。 25 mm, length = 6 cm) was prepared in a machine shop and used to (中田石無故片(D=4 mm)、长度=5 mm)和磁化钨圆柱杆 使用石墨垫片(D=4 mm,长度=5 mm)和碳化钨圆柱杆 load the reactant powder Graphite spacers (D = 4 mm, length = 5 (D=4 mm, 长度=5 cm)作为电极来压缩加载的粉末。 mm) and tungsten carbide cylinder rods (D = 4 mm, length = 5 cm) were utilized as electrodes to compress the loaded powder. The 石墨垫片与样品接触,碳化钨圆柱杆与电路连接。 graphite spacers were in contact with the samples and tungsten 在这项工作甲一使 在这项工作甲一使 和不同的前躯体来制备掺杂杂原子的FG。 different precursors were used to prepare heteroatom doped FG. The 反应物包括(1)炭黑。(2)三聚酮酸和炭黑。(3)硼酸和炭黑。(4)蔗糖和炭黑。(5) reactants included a mixture of (1) CB, (2) melamine and CB, (3) 红磷和炭黑。(6)聚苯硫酸和炭黑、(7)硼。三聚酮酸和发黑和合物、以及(8)硼、三 boric acid and CB, (4) sucrose and CB, (5) red phosphorus and CB, 聚氰酸、聚苯硫酸和发展的混合物。 melamine, PPS, and CB. The electrical energy was provided by a capacitor bank in the circuit with a total capacitance of 60 mF. The 电容器组由可达到400 V的申源直流充曲 电容器组由可达到400 V的电源直流充电。 capacitor bank was charged by a d.c. supply that could reach 400 V. 闪泰持续时间由电路中充当高速开关的Arduino控制器继电器控制。 The flash duration was controlled by an Arduino controller relay in 不同反应物的反应物电阻不同,改变闪 TPI风巡视时风巡视时间, 战毁风 在时间 as a high-speed switch. The reactant resistances for 然条件以控制忌物入电热能。 various reactants were different, and the flash conditions were キスセイトレビット changed to control the total input electrothermal energy. To scale up N-FG-L1使用了厚壁聚合物管(ID=8mm, OD=25mm, 长度=6cm)和石墨垫片(D=8mm, 长度 the N-FG, thick-walled polymer tube (ID = 8 mm, OD = 25 mm, =5mm)。 length = 6 cm) and graphite spacers (D = 8 mm, length = 5 mm) were 石英管(內径=16 mm, 外径=20 mm, 长度=6 cm)和石墨垫片 used for N-FG-L1. The quartz tube (ID = 16 mm, OD = 20 mm, (首径=16 mm, 长度=20 mm, 大度=20 mm, calculated based on the energy densities in Table S1. The parameters 表优化了各种样品的重量比。 are listed in Table 1. The weight ratios of various samples are 我们试图优化基络条件的现入因素: optimized in Table 1. There are four factors for which we try to optimize the final conditions:

均匀闪蒸反应。 我们将均匀混合的反应物的电阻控制在0.4到 1. Homogeneous flash reactions. We control the resistance of the evenly mixed reactants between 0.4–5.0  $\Omega$ , and monitor the

确保均匀闪蒸反应的电流·时间(I-t)曲线 current—time (I-t) curves to ensure a homogeneous flash reaction

- 涡轮层状石墨烯的形成。 <u>
  拉曼光谱和X射线衍射用于检查</u> 2. <u>
  The formation of turbo</u>stratic graphene. Raman and XRD are 海轮层句壑炳的形成,以师孤东针。 used to check the formation of turbostratic graphene to screen the condition
- $sp_2$ -C和杂原子之间键的形成。 3. The formation of bonds between  $sp_2^2$ -C and heteroatom. Since 由于目的是制备掺杂石墨筛,因此应排陈\$P3-G和采原于《问键百时》则能注。 the purpose is to prepare doped graphene, the possibility of bonds between sp<sup>3</sup>-C and heteroatom should be excluded. XPS is used to analyze the bonding states.
- 杂原子的含量。 XPS用于分析杂原子的含量。 4. The contents of the heteroatom. XPS is used to analyze the contents of the heteroatom.

f 新 Line and the letteroatom. 有 新 Line and a circuit diagram of the FJH setup are in the Line and a circuit diagram of the FJH setup are in the Line and a circuit diagram of the FJH setup are in the Line and the the state of the state of the setup are in the Line and the state of the state Line and the state of the state and the state of the sta 裔,开且电容器组已元至放电。 separated from the system by a circuit breaker (not a toggle switch), and that the capacitor bank has been fully discharged.

単化字測试。 **Electrochemical Tests**. Oxygen reduction reaction (ORR). The 週2147 mg采原于掺杂的质性之机后有加溶液中制备采料,该粘合加溶液为160 5 %% Nation溶 slurry was prepared by dispersing 2 mg heteroatom doped FG in 2 mL 流 混合在2 ml为(7 葩 14) 体和比 恤 波, 混合任2012年(11, ) 件気に ) で。 binder solution, which is 160 µL of 5 wt % Nafion solution mixed in 2 將恣料超声处理30分钟以形成均匀的油墨。 mL water/ethanol (1:1, volume ratio). 将浆料超声处理30分钟以形成均匀的油量。 30 min to form a homogeneous ink. A volume of 20 µL of ink 电极(直径为5mm)上,并在室温下在空气中干叠。 电极(直径为5mm)上,并在室温下在空气中干燥。 the A volume of 20 µL of ink solution was loaded onto a glassy carbon electrode (5 mm in 使用CH1600B电化学工作站 diameter) and dried in air at room temperature. The electrochemical 在三电极局置中进行电化学测量。 在二电极配置中进行电化子测量。 measurements were carried out in a three-electrode configuration 石黒様和HoO/Hol 1 M 石墨棒和HgO/Hg1M KOH 角极分别用作计数器和参计曲部 KOH 博威分開用作計数器和参比电极。 HgO/Hg (1 M KOH) electrode were used as the counter and 电解液为(1 M KOH) electrode were used as the counter and neference electrodes, respectively. The electrolyte was 0.1 M KOHsparged with O, gas for O, saturation. CV was carried out with 0/(vsRHE)的电位范围内进行(0, - 周期) by 0.2 Was carried out with potential ranging from 1.2 to 0 V (vs RHE) for 10 cycles at a scan rate of 100 mV s<sup>-1</sup>. LSV was carried out with potential ranging from 1.2 to 0V(vsRHE), 扫描速率为5mV/s。 0 V (vs RHE) at a scan rate of 5 mV s<sup>-1</sup> 在电池测试中,通过0.90.10H比例研磨掺杂FG和聚二氟乙烯(PVDF)的混合物来制备电极。 For the battery tests, the electrode was prepared by grinding the

mixture of doped FG, and poly(vinyl difluoride) (PVDF) at a ratio of 使用少量(总质量的3倍)N-用基吡咯烷酮形成均匀源料。 0.9:0.1. A small amount (~3× of the total mass) of N-methyl pyrrolidone was used to form a homogeneous slurry. The slurries 大技廠仍公验形成並述 下球層10万世形成洗液。 were formed by ball milling at 1500 rpm for 10 min. The current 10 µm的領資。 collector was Cu foil with a thickness of 10 µm. The slurry was 領資上, 刮墨刀间距为100µm。 applied to the Cu foil by a doctor blade with a blade spacing of 100 使用放着在中級问题的內面加热為在70°C下干燥电极20时,然后将其防人真空炉中过夜。 使用放置在电极则部的内置加热盖在 $10^{\circ}$ C下十燥电极20时,然后将其放入其至炉中过夜。  $\mu$ m. The electrode was dried using a built-in heating cover placed on top of the electrode at 70 °C for 2 h and was then put in a vacuum  $\underline{A}^{\text{pressure}}_{\text{pressure}}$  oven overnight. The temperature and pressure of the vacuum oven  $\underline{B}^{\text{pressure}}_{\text{pressure}}$  of th 为深加剂。 DOL (V:V = 1:1) with 2% LiNO<sub>3</sub> as the additive. The volume of the electrolyte in each coin cell was 30  $\mu$ L.

#### ASSOCIATED CONTENT

# Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsnano.2c01136.

Safety instructions, TEM, SEM, and other images; XPS and Raman spectra; Additional graphs (PDF)

#### **AUTHOR INFORMATION**

## **Corresponding Author**

James M. Tour - Department of Chemistry, Department of Materials Science and NanoEngineering, Smalley-Curl Institute, the NanoCarbon Center, and The Welch Institute for Advanced Materials, and Department of Computer Science, Rice University, Houston, Texas 77005, United States; O orcid.org/0000-0002-8479-9328; Email: tour@ rice.edu

#### Authors

ACS Nano

- Weiyin Chen Department of Chemistry, Rice University, Houston, Texas 77005, United States
- Chang Ge Department of Chemistry and Applied Physics Program, Rice University, Houston, Texas 77005, United States
- John Tianci Li Department of Chemistry and Department of Materials Science and NanoEngineering, Rice University, Houston, Texas 77005, United States; © orcid.org/0000-0002-7218-8298
- Jacob L. Beckham Department of Chemistry, Rice University, Houston, Texas 77005, United States
- **Zhe Yuan** Department of Chemistry, Rice University, Houston, Texas 77005, United States
- Kevin M. Wyss Department of Chemistry, Rice University, Houston, Texas 77005, United States
- Paul A. Advincula Department of Chemistry, Rice University, Houston, Texas 77005, United States
- Lucas Eddy Department of Chemistry and Applied Physics Program, Rice University, Houston, Texas 77005, United States
- Carter Kittrell Department of Chemistry, Rice University, Houston, Texas 77005, United States; Ocicid.org/0000-0002-8449-4292
- Jinhang Chen Department of Chemistry, Rice University, Houston, Texas 77005, United States
- Duy Xuan Luong Department of Chemistry, Rice University, Houston, Texas 77005, United States; Occid.org/0000-0001-7089-3359
- Robert A. Carter Department of Chemistry, Rice University, Houston, Texas 77005, United States

Complete contact information is available at: https://pubs.acs.org/10.1021/acsnano.2c01136

## **Author Contributions**

<sup>#</sup>W.C. and C.G. contributed equally to this work.

## Notes

The authors declare the following competing financial interest(s): Rice University owns intellectual property on flash graphene. Some of that intellectual property has been licensed to Universal Matter Inc. J.M.T. is a stockholder in Universal Matter, but he is not an officer, director, or employee. Conflicts of interest are mitigated through regular disclosures to and compliance with the Rice University Office of Sponsored Programs and Research Compliance. D.X.L. was employed by Rice University when this work was done. He is now employed by Universal Matter.

#### ACKNOWLEDGMENTS

The funding of the research is provided by Air Force Office of Scientific Research (FA9550-19-1-0296), DOE-NETL (DE-FE0031794), and U.S. Army Corps of Engineers, ERDC (W912HZ-21-2-0050). The characterization equipment used in this project is from the Shared Equipment Authority (SEA) at Rice University. The authors acknowledge the use of the Electron Microscopy Center (EMC) at Rice University. We thank Dr. Bo Chen for helpful discussion of the XPS results, and Dr. Chris Pennington for method development of the ICP-MS tests. We thank Prof. Yufeng Zhao for helpful discussion and suggestion of the experiments.

# REFERENCES

(1) Agnoli, S.; Favaro, M. Doping Graphene with Boron: A Review of Synthesis Methods, Physicochemical Characterization, and Emerging Applications. J. Mater. Chem. A **2016**, *4*, 5002–5025.

(2) Xu, D.; Chen, W.; Zeng, M.; Xue, H.; Chen, Y.; Sang, X.; Xiao, Y.; Zhang, T.; Unocic, R. R.; Xiao, K.; Fu, L. Crystal-Field Tuning of Photoluminescence in Two-Dimensional Materials with Embedded Lanthanide Ions. *Angew. Chem., Int. Ed.* **2018**, *57*, 755–759.

(3) Peng, Z.; Ye, R.; Mann, J. A.; Zakhidov, D.; Li, Y.; Smalley, P. R.; Lin, J.; Tour, J. M. Flexible Boron-Doped Laser-Induced Graphene Microsupercapacitors. *ACS Nano* **2015**, *9*, 5868–5875.

(4) Ding, Y.; Zeng, M.; Zheng, Q.; Zhang, J.; Xu, D.; Chen, W.; Wang, C.; Chen, S.; Xie, Y.; Ding, Y.; Zheng, S.; Zhao, J.; Gao, P.; Fu, L. Bidirectional and Reversible Tuning of the Interlayer Spacing of Two-Dimensional Materials. *Nat. Commun.* **2021**, *12*, 5886.

(5) Wang, X.; Sun, G.; Routh, P.; Kim, D.-H.; Huang, W.; Chen, P. Heteroatom-Doped Graphene Materials: Syntheses, Properties and Applications. *Chem. Soc. Rev.* **2014**, *43*, 7067–7098.

(6) Wu, G.; McHugh, E. A.; Berka, V.; Chen, W.; Wang, Z.; Beckham, J. L.; Derry, P. J.; Roy, T.; Kent, T. A.; Tour, J. M.; Tsai, A.-L. Oxidized Activated Charcoal Nanoparticles as Catalytic Superoxide Dismutase Mimetics: Evidence for Direct Participation of an Intrinsic Radical. *ACS Appl. Nano Mater.* **2020**, *3*, 6962–6971.

(7) Advincula, P. A.; Luong, D. X.; Chen, W.; Raghuraman, S.; Shahsavari, R.; Tour, J. M. Flash Graphene from Rubber Waste. *Carbon* **2021**, *178*, 649–656.

(8) Lowell, C. E. Solid Solution of Boron in Graphite. J. Am. Ceram. Soc. **1967**, 50, 142–144.

(9) Yu, X.; Han, P.; Wei, Z.; Huang, L.; Gu, Z.; Peng, S.; Ma, J.; Zheng, G. Boron-Doped Graphene for Electrocatalytic  $N_2$  Reduction. *Joule* **2018**, *2*, 1610–1622.

(10) Zhang, R.; Chen, X.-R.; Chen, X.; Cheng, X.-B.; Zhang, X.-Q.; Yan, C.; Zhang, Q. Lithiophilic Sites in Doped Graphene Guide Uniform Lithium Nucleation for Dendrite-Free Lithium Metal Anodes. *Angew. Chem., Int. Ed.* **2017**, *56*, 7764–7768.

(11) Han, J.; Cheon, J. Y.; Joo, S. H.; Park, S. Synthesis of Boron and Nitrogen Co-Doped Graphene Nano-Platelets Using a Two-Step Solution Process and Catalytic Properties for Oxygen Reduction Reaction. *Solid State Sci.* **2014**, *33*, 1–5.

(12) Chen, H.; Luo, Q.; Liu, T.; Tai, M.; Lin, J.; Murugadoss, V.; Lin, H.; Wang, J.; Guo, Z.; Wang, N. Boosting Multiple Interfaces by Co-Doped Graphene Quantum Dots for High Efficiency and Durability Perovskite Solar Cells. *ACS Appl. Mater. Interfaces* **2020**, *12*, 13941–13949.

(13) Sheng, Z.-H.; Gao, H.-L.; Bao, W.-J.; Wang, F.-B.; Xia, X.-H. Synthesis of Boron Doped Graphene for Oxygen Reduction Reaction in Fuel Cells. *J. Mater. Chem.* **2012**, *22*, 390–395.

(14) Mu, X.; Yuan, B.; Feng, X.; Qiu, S.; Song, L.; Hu, Y. The Effect of Doped Heteroatoms (Nitrogen, Boron, Phosphorus) On Inhibition Thermal Oxidation of Reduced Graphene Oxide. *RSC Adv.* **2016**, *6*, 105021–105029.

(15) Panchakarla, L. S.; Subrahmanyam, K. S.; Saha, S. K.; Govindaraj, A.; Krishnamurthy, H. R.; Waghmare, U. V.; Rao, C. N. R. Synthesis, Structure, and Properties of Boron- and Nitrogen-Doped Graphene. *Adv. Mater.* **2009**, *21*, 4726–4730.

(16) Tang, Y.-B.; Yin, L.-C.; Yang, Y.; Bo, X.-H.; Cao, Y.-L.; Wang, H.-E.; Zhang, W.-J.; Bello, I.; Lee, S.-T.; Cheng, H.-M.; Lee, C.-S. Tunable Band Gaps and p-Type Transport Properties of Boron-Doped Graphenes by Controllable Ion Doping Using Reactive Microwave Plasma. *ACS Nano* **2012**, *6*, 1970–1978.

(17) Wu, T.; Shen, H.; Sun, L.; Cheng, B.; Liu, B.; Shen, J. Nitrogen and Boron Doped Monolayer Graphene by Chemical Vapor Deposition Using Polystyrene, Urea and Boric Acid. *New J. Chem.* **2012**, *36*, 1385–1391.

(18) Zhou, Z.; Wakamiya, A.; Kushida, T.; Yamaguchi, S. Planarized Triarylboranes: Stabilization by Structural Constraint and Their Plane-to-Bowl Conversion. *J. Am. Chem. Soc.* 2012, *134*, 4529–4932.
(19) Luong, D. X.; Bets, K. V.; Algozeeb, W. A.; Stanford, M. G.; Kittrell, C.; Chen, W.; Salvatierra, R. V.; Ren, M.; McHugh, E. A.;

Advincula, P. A.; Wang, Z.; Bhatt, M.; Guo, H.; Mancevski, V.; Shahsavari, R.; Yakobson, B. I.; Tour, J. M. Gram-Scale Bottom-Up Flash Graphene Synthesis. *Nature* **2020**, *577*, 647–651.

(20) Chen, W.; Wang, Z.; Bets, K. V.; Luong, D. X.; Ren, M.; Stanford, M. G.; McHugh, E. A.; Algozeeb, W. A.; Guo, H.; Gao, G.; Deng, B.; Chen, J.; Li, J. T.; Carsten, W. T.; Yakobson, B. I.; Tour, J. M. Millisecond Conversion of Metastable 2D Materials by Flash Joule Heating. ACS Nano 2021, 15, 1282–1290.

(21) Algozeeb, W. A.; Savas, P. E.; Luong, D. X.; Chen, W.; Kittrell, C.; Bhat, M.; Shahsavari, R.; Tour, J. M. Flash Graphene from Plastic Waste. *ACS Nano* **2020**, *14*, 15595–15604.

(22) Yao, Y.; Huang, Z.; Xie, P.; Lacey, S. D.; Jacob, R. J.; Xie, H.; Chen, F.; Nie, A.; Pu, T.; Rehwoldt, M.; Yu, D.; Zachariah, M. R.; Wang, C.; Shahbazian-Yassar, R.; Li, J.; Hu, L. Carbothermal Shock Synthesis of High-Entropy-Alloy Nanoparticles. *Science* **2018**, 359, 1489–1494.

(23) Li, T.; Pickel, A. D.; Yao, Y.; Chen, Y.; Zeng, Y.; Lacey, S. D.; Li, Y.; Wang, Y.; Dai, J.; Wang, Y.; Yang, B.; Fuhrer, M. S.; Marconnet, A.; Dames, C.; Drew, D. H.; Hu, L. Thermoelectric Properties and Performance of Flexible Reduced Graphene Oxide Films Up To 3,000 K. *Nat. Energy* **2018**, *3*, 148–156.

(24) Chen, W.; Li, J. T.; Wang, Z.; Algozeeb, W. A.; Luong, D. X.; Kittrell, C.; McHugh, E. A.; Advincula, P. A.; Wyss, K. M.; Beckham, J. L.; Stanford, M. G.; Jiang, B.; Tour, J. M. Ultrafast and Controllable Phase Evolution by Flash Joule Heating. *ACS Nano* **2021**, *15*, 11158– 11167.

(25) Iglesias, D.; Giuliani, A.; Melchionna, M.; Marchesan, S.; Criado, A.; Nasi, L.; Bevilacqua, M.; Tavagnacco, C.; Vizza, F.; Prato, M.; Fornasiero, P. N-Doped Graphitized Carbon Nanohorns as a Forefront Electrocatalyst in Highly Selective  $O_2$  Reduction to  $H_2O_2$ . *Chem.* **2018**, *4*, 106–123.

(26) Pollard, A. J.; Brennan, B.; Stec, H.; Tyler, B. J.; Seah, M. P.; Gilmore, I. S.; Roy, D. Quantitative Characterization of Defect Size in Graphene Using Raman Spectroscopy. *Appl. Phys. Lett.* **2014**, *105*, 253107.

(27) Jorio, A.; Ferreira, E. H. M.; Moutinho, M. V. O.; Stavale, F.; Achete, C. A.; Capaz, R. B. Measuring Disorder in Graphene with the G and D Bands. *Phys. Status Solidi B* **2010**, *247*, 2980–2982.

(28) Stanford, M. G.; Bets, K. V.; Luong, D. X.; Advincula, P. A.; Chen, W.; Li, J. T.; Wang, Z.; McHugh, E. A.; Algozeeb, W. A.; Yakobson, B. I.; Tour, J. M. Flash Graphene Morphologies. *ACS Nano* **2020**, *14*, 13691–13699.

(29) Tokarczyk, M.; Kowalski, G.; Witowski, A.M.; Kozinski, R.; Librant, K.; Aksienionek, M.; Lipinska, L.; Ciepielewski, P. Structural and Electronic Properties of Graphene Oxide and Reduced Graphene Oxide Papers Prepared by High Pressure and High Temperature Treatment. *Acta Phys. Polym., A* **2014**, *126*, 1190–1194.

(30) Gladkaya, I. S.; Kremkova, G. N.; Slesarev, V. N. Turbostratic Boron Nitride (BN<sub>t</sub>) Under High Pressure and Temperature. *J. Less-Common Met.* **1986**, *117*, 241–245.

(31) Thomas, J., Jr.; Weston, N. E.; O'Connor, T. E. Turbostratic Boron Nitride, Thermal Transformation to Ordered-layer-lattice Boron Nitride. *J. Am. Chem. Soc.* **1962**, *84*, 4619–4622.

(32) Inagaki, M.; Kang, F. Materials Science and Engineering of Carbon: Fundamentals, 2nd ed.; Elsevier Inc., 2014; pp 17–217. DOI: 10.1016/C2013-0-13699-9.

(33) Franklin, R. E. Crystallite Growth in Graphitizing and Non-Graphitizing Carbons. *Proc. R. Soc. London A* 1951, 209, 196–218.
(34) Ouzilleau, P.; Gheribi, A. E.; Chartrand, P.; Soucy, G.;

Monthioux, M. Why Some Carbons May or May Not Graphitize? The Point of View of Thermodynamics. *Carbon* **2019**, *149*, 419–435.

(35) Nakanishi, K. Porosity Measurement. Handbook of Sol-Gel Science and Technology. *Springer International Publishing AG* **2016**, 1–11.

(36) Radlinski, A. P.; Mastalerz, M.; Hinde, A. L.; Hainbuchner, M.; Rauch, H.; Baron, M.; Lin, J. S.; Fan, L.; Thiyagarajan, P. Application of SAXS and SANS in Evaluation of Porosity, Pore Size Distribution and Surface Area of Coal. *Int. J. Coal Geol.* **2004**, *59*, 245–271. (37) Chen, W.-T.; Dutta, D.; Hung, Y.-H.; Sin, Y.-Y.; He, S.-M.; Chang, J.-K.; Su, C.-Y. Designed Catalytic Protocol for Enhancing Hydrogen Evolution Reaction Performance of P, N-Co-Doped Graphene: The Correlation of Manipulating the Dopant Allocations and Heteroatomic Structure. *J. Phys. Chem. C* **2020**, *124*, 25701– 25711.

(38) Sibul, R.; Põldsepp, E. K.; Mäeorg, U.; Merisalu, M.; Kikas, A.; Kisand, V.; Treshchalov, A.; Sammelselg, V.; Tammeveski, K. Sulphur and Nitrogen Co-Doped Graphene-Based Electrocatalysts for Oxygen Reduction Reaction In Alkaline Medium. *Electrochem. Commun.* **2019**, *109*, 106603.

(39) Wyss, K. M.; Beckham, J. L.; Chen, W.; Luong, D. X.; Hundi, P.; Raghuraman, S.; Shahsavari, R.; Tour, J. M. Converting Plastic Waste Pyrolysis Ash into Flash Graphene. *Carbon* **2021**, *174*, 430– 438.

(40) Wang, Z.; Li, Q.-K.; Zhang, C.; Cheng, Z.; Chen, W.; McHugh, E. A.; Carter, R. A.; Yakobson, B. I.; Tour, J. M. Hydrogen Peroxide Generation with 100% Faradaic Efficiency on Metal-Free Carbon Black. ACS Catal. **2021**, *11*, 2454–2459.

(41) Wang, L.; Sofer, Z.; Pumera, M. Will Any Crap We Put into Graphene Increase Its Electrocatalytic Effect? *ACS Nano* **2020**, *14*, 21–25.

(42) Ye, R.; Dong, J.; Wang, L.; Cruz, R. M.; Li, Y.; An, P.-F.; Yacaman, M. J.; Yakobson, B. I.; Chen, D.; Tour, J. M. Manganese Deception on Graphene and Implications in Catalysis. *Carbon* **2018**, *132*, 623–631.

(43) Wang, L.; Ambrosi, A.; Pumera, M. "Metal-Free" Catalytic Oxygen Reduction Reaction on Heteroatom-Doped Graphene is Caused by Trace Metal Impurities. *Angew. Chem., Int. Ed.* **2013**, *52*, 13818–13821.

(44) Chen, W.; Salvatierra, R. V.; Ren, M.; Chen, J.; Stanford, M. G.; Tour, J. M. Laser-Induced Silicon Oxide for Anode-Free Lithium Metal Batteries. *Adv. Mater.* **2020**, *32*, 2002850.

(45) Salvatierra, R. V.; Chen, W.; Tour, J. M. What Can be Expected from "Anode-Free" Lithium Metal Batteries? *Adv. Energy Sustainability Res.* **2021**, *2*, 2000110.

(46) For scaleup of FJH for graphene synthesis, see: https://www.universalmatter.com/ (accessed December 8, 2021).